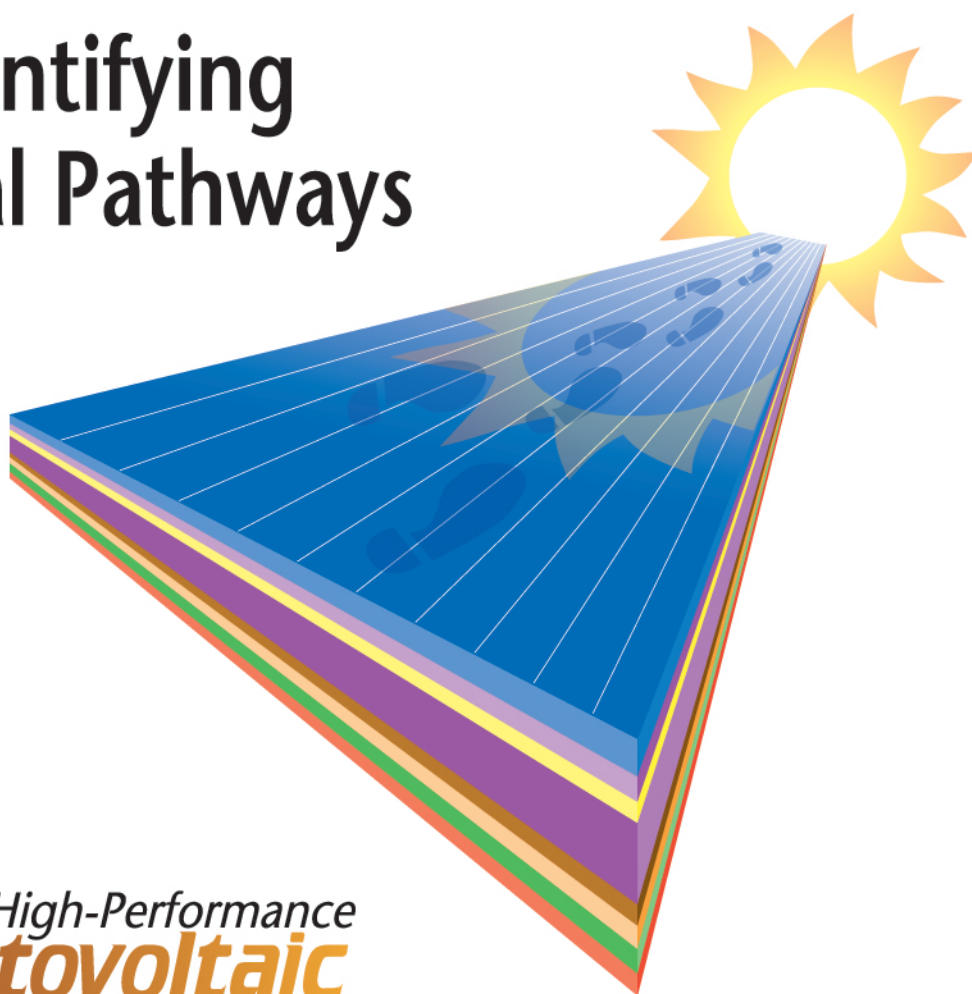


High-Performance Photovoltaic Project

Kickoff Meeting

NREL • October 18, 2001

Identifying Critical Pathways



NREL / BK-520-31284

High-Performance Photovoltaic Project

Kickoff Meeting
October 18, 2001

**Identifying
Critical Pathways**



National Renewable Energy Laboratory
1617 Cole Boulevard
Golden, Colorado 80401-3393

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KICKOFF MEETING AGENDA

- 8:00-8:10 Martha Symko-Davies, National Renewable Energy Laboratory
Welcome
- 8:10-8:25 Tim Coutts, National Renewable Energy Laboratory
"Realistic Modeling of Thin-Film Tandem Solar Cells"
- 8:25-8:40 William Shafarman, University of Delaware
"Wide Band Gap Cu(InGa)(SeS)₂ and (CdZn)Te Thin Films for Tandem Solar Cells"
- 8:40-8:55 David Young, National Renewable Energy Laboratory
"NCPV Thin-Film Tandem Research"
- 8:55-9:10 Akhlesh Gupta, University of Toledo
"II-VI Tunnel Junctions and Absorber Alloys by Magnetron Sputtering"
- 9:10-9:25 Ingrid Eisgruber, Global Solar
"Progress Toward 20% Efficient CuIn_xGa_{1-x}Se₂ Photovoltaic Devices on Foil Substrates"
- 9:25-9:40 Doug Rose, First Solar LLC
"High Performance PV Tasks at First Solar"
- 9:40-9:55 Chris Ferekides, University of South Florida
"II-VI Based High Band Gap Devices for Tandem Applications"
- 9:55-10:10 Oscar Crisalle, University of Florida
"CGS/CIGS Tandem Cells: Critical Manufacturing Issues"
- 10:10-10:25 Break
- 10:25-10:40 Angus Rockett, University of Illinois
"CuInSe₂ Heterojunctions and Heterojunction Solar Cells With GaAs and Ge"
- 10:40-10:55 Michael Mauk, AstroPower, Inc.
"InGaP/GaAs-on-Ceramic Thin-Film Monolithically Interconnected, Large Area, Tandem Solar Cell Array"
- 10:55-11:10 Sarah Kurtz, National Renewable Energy Laboratory
"NREL Basic Research Toward a 40% Efficiency"
- 11:10-11:25 Keith Emery, National Renewable Energy Laboratory
"Reference Conditions for PV Concentrators"

- 11:25-11:40 Raed Shariff, Spectrolab, Inc.
"High Performance, Low Cost III-V PV Concentrator Module"
- 11:40-11:55 Mark O'Neil, ENTECH, Inc.
"Development of Terrestrial Concentrator Modules Using High-Efficiency Multi-Junction Solar Cells"
- 11:55-12:10 Mark Stan, Emcore
"A Three-Junction Solar Cell for High Concentration Applications".
- 12:10-12:30 Break
- 12:30-1:30 Discussion

Topics:

- 1) Sharing/coordinating of information
- 2) Website
- 3) Logistics - Meeting, when, where, and how often
- 4) Report at end of Phase I
- 5) Nonstructured group meetings - inclusive for next phase??

High-Performance Photovoltaic Project

Kickoff Meeting

Identifying Critical Pathways

Martha Symko-Davies

National Center for Photovoltaics • National Renewable Energy Laboratory

October 18, 2001



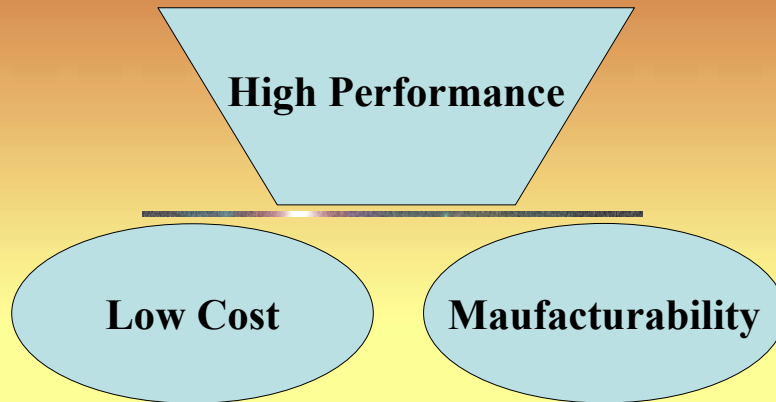
Near Term Key Targets	Date
Demonstrate a 20% Efficiency Thin Film Cell under Low Concentration	2001
Identify Key Issues and Pathways for Achieving a 33% Concentrator Module and a 25% Thin Film Multijunction Cell	2002
Demonstrate a 34% Cell under Concentration	2003
Full Implementation of Thin Film Process Integration	2004
Fabricate a Dual-Junction Thin Film polycrystalline Cell of 15% efficiency	2005

Demonstrate
Polycrystalline Dual-Junction
Thin Film Cell of 25% Efficiency

Demonstrate a
40% Cell under Concentration



What Makes this Program Different??



Purpose of Meeting????

Avenue to meet and introduce yourself to others in this project

Basis for **sharing** information and **collaboration**

Discussion Topics





The screenshot shows the 'High-Performance Photovoltaic Project' website. At the top is a banner with a solar panel image and the project name. Below the banner is a navigation menu on the left with links: Overview, Working with Industry, Our Partners, Publications, News and Events, Message Board, and Contacts. The main content area features several images: a 3D diagram of a multi-junction solar cell, a photograph of a solar array on a building, a network diagram with nodes, and a diagram of a tandem solar cell structure with layers labeled 'GaAs', 'Top cell absorber', 'Bottom cell absorber', and 'Substrate'.

<http://www.nrel.gov/highperformancepv>




High Performance PV subcontractors Polycrystalline

Subcontractor	Title
*Astropower	InGaP/GaAs-on Ceramic Thin-Film Monolithically Interconnected, Large Area, Tandem Solar Cell Array
University of Delaware	Thin Film Multijunction Solar Cells: Development of a High Bandgap Cell (CIS and CdTe alloys)
University of Florida	Identification of Critical Paths in the Manufacturing of Low-Cost High-Efficiency CGS/CIS Two-Junction Tandem Cells (Single-Crystal CIS-alloy top cells on GaAs)
University of Toledo	Polycrystalline Thin-Film Tandem Photovoltaic Cells (CdTe-alloys)
University of South Florida	Development of a II-VI-Based High Performance, High Band Gap Device for Thin-Film Tandem Solar Cells(CdSe and CdTe-alloy)
Global Solar	Progresss Toward 20% Efficient $\text{CuIn}_x\text{Ga}_{1-x}\text{Se}_2$ Photovoltaic Devices on Foil Substrates



High Performance PV subcontractors Concentrators

Subcontractor	Title
Univ. of Illinois	Cu(In,Ga)Se ₂ Heterojunction Solar Cells for Extreme High-efficiency Photovoltaic Concentrators (CIS on single-crystal GaAs for potential use in III-V multijunctions for concentrators)
Entech, Inc.	Near-Term Integration of III-V Cells Operating at 440X, Into Entech's Field Proven Concentrator Module
SunPower Corporation	Lens-Based Concentrator Modules: Exploring Critical Optical and System Integration Issues
Spectrolab, Inc.	High Efficiency, Low Cost, III-V Concentrator PV Cell & Receiver Module
Emcore	A Three-Junction Solar Cell for High Concentration Applications (III-V multijunctions-lattice mismatched structure)



Discussion Topics

- 1) Purpose of Meetings
- 2) Structure: Working Groups???
- 3) Meeting Logistics -when, where, and how often
- 4) Report at end of Phase I
- 5) Inclusive for next phase



Realistic modeling of thin-film, tandem solar cells

Timothy J. Coutts, J. Scott Ward,
David L. Young, Timothy A. Gessert, and
Rommel Noufi

tim_coutts@nrel.gov



*NREL

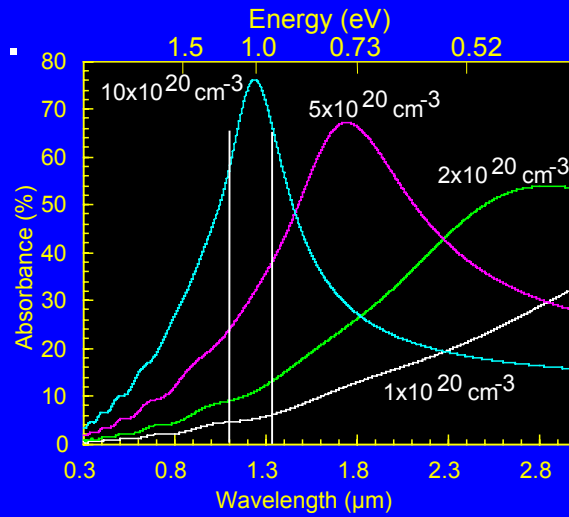


Issues

- A transparent conducting oxide (TCO) is used in all thin-film solar cells
- Free-electrons absorb in the near-infrared portion of the spectrum
- Depending on their concentration and mobility, they can impact cell performance
- In tandem cells their effect can be large

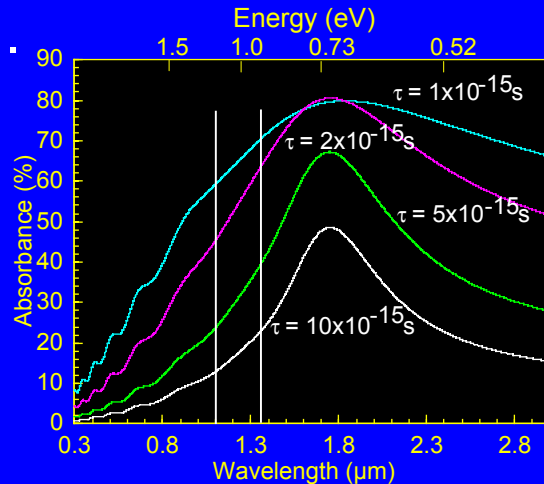
Dependence of TCO absorbance on free-carrier concentration

- $t = 500 \text{ nm}$
- $\epsilon_{\infty} = 4.4$
- $m^* = 0.35 m_e$
- $\tau = 5 \times 10^{-15} \text{ s}$
- $\mu = 25 \text{ cm}^2 \text{ s}^{-1} \text{ V}^{-1}$



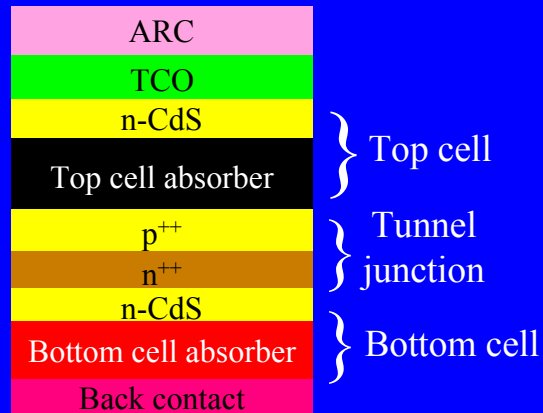
Dependence of absorbance on free-carrier relaxation time

- $t = 500 \text{ nm}$
- $n = 5 \times 10^{20} \text{ cm}^{-3}$
- $\epsilon_{\infty} = 4.4$
- $m^* = 0.35 m_e$
- $\mu = 5-50 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$



Schematic of modeled thin-film tandem cell

- All interfaces assumed to be specular
- No interdiffusion
- Top p-type absorber assumed to be chalcopyrite



Assumptions and modeling approach

- Calculate light transmitted into the top (chalcopyrite) absorber, and, hence, J_{sc1} (Global reference spectrum)
- Assume all photons with $E_{g1} < E < E_{g2}$ contribute to J_{sc2}
- Put J_{sc} equal to the smaller of J_{sc1} and J_{sc2}
- Take ideality factor as 1.5
- Calculate J_0 from a quasi-empirical model
- Calculate $V_1(J)$, $V_2(J)$ and $V_T(J)$
- Calculate fill-factor, P_{mpp} and efficiency as $f(E_{g1}, E_{g2})$

Modeling of J_0

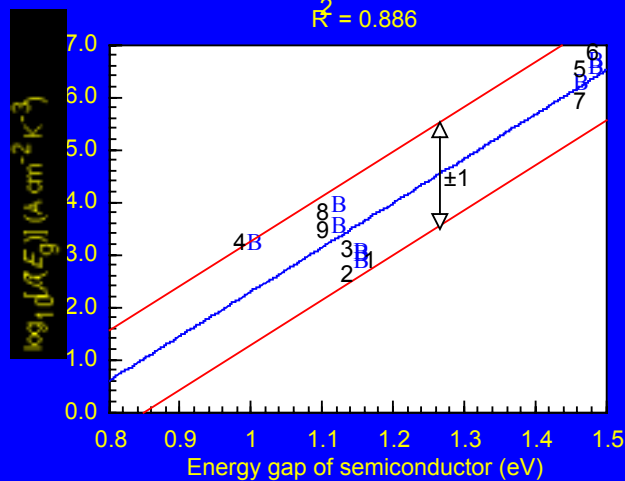
$$J_0(E_g) = \beta(E_g) T^3 \exp(-E_g/kT)$$

- Calculate $\beta(E_g)$ and correlate with E_g for the best single-junction thin-film devices
- This gives a reasonable estimate of $J_0(E_g)$ for devices of uninvestigated bandgaps
- It does **not** define the absolute limit

$\text{Log}_{10}\beta(E_g)$ vs. E_g

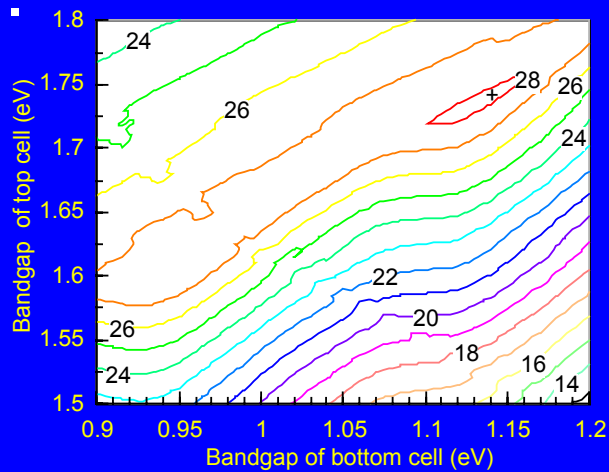
$$\beta(E_g) = 7.1579 \times 10^7 \times \text{Exp}[19.4274 E_g]$$

$$R^2 = 0.886$$



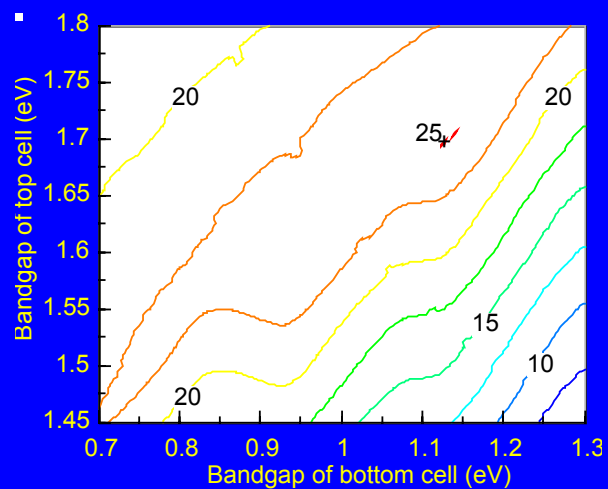
Modeled tandem efficiency (i)

- Assumes all photons reach the top absorber
- Uses central correlation line
- Absolute maximum is 28.2%



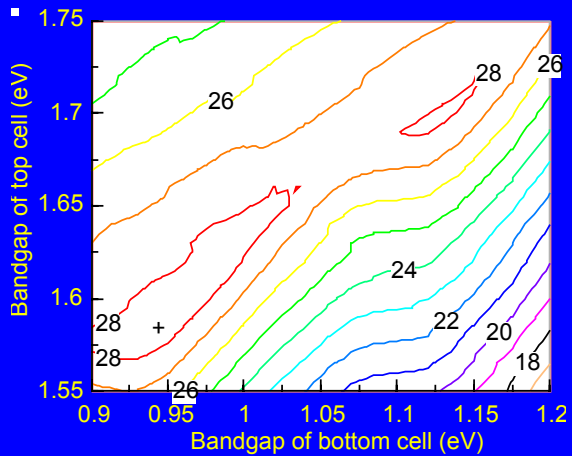
Modeled tandem efficiency (ii)

- TCO thickness = 500 nm
- $n = 1 \times 10^{20} \text{ cm}^{-3}$
 $\mu = 75 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$
- CdS thickness = 50 nm
- MgF_2 thickness = 100 nm
- Maximum efficiency = 25.0%



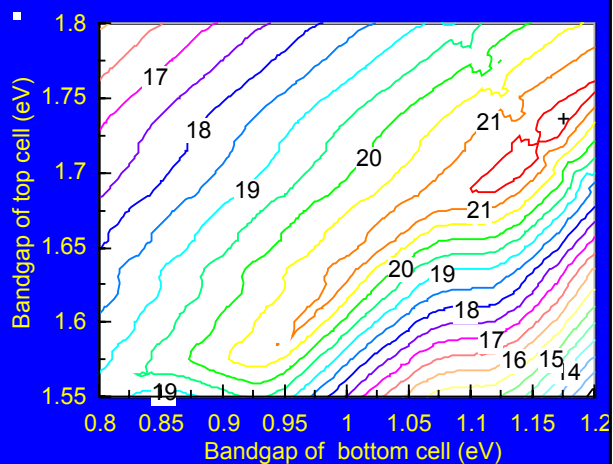
Modeled tandem efficiency (iii)

- Uses the same front layers
- Uses the lowest of the three lines in the $\log \beta(E_g)$ correlation
- Maximum efficiency = 28.7%



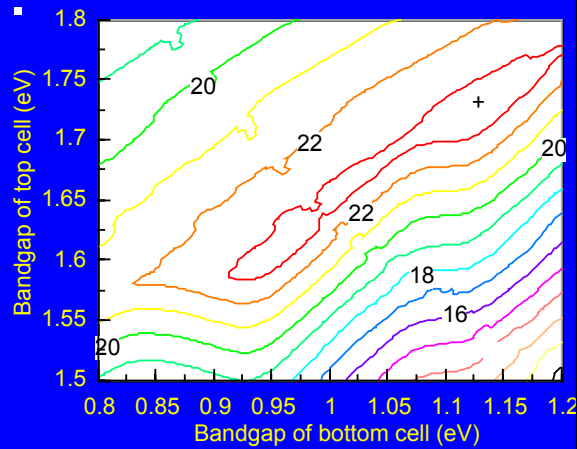
Modeled tandem efficiency (iv)

- Uses the same front layers
- Uses the upper-most of the three lines in the $\log \beta(E_g)$ line
- Maximum efficiency = 21.8%



Modeled tandem efficiency (v)

- Uses a TCO with $n = 5 \times 10^{20} \text{ cm}^{-3}$, $\mu = 75 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, $t = 500 \text{ nm}$
- Uses central $\log \beta(E_g)$ line
- Maximum efficiency = 23.6%



Summary

- TCOs with low carrier concentration and high mobility needed to minimize free-carrier absorption
- Beneficial to eliminate the TCO and the CdS
- Depending on J_0 , the optimum bandgap pair is about 1.7 and 1.13 eV
- The goal of 25% is achievable but challenging!
- Reduction in J_0 would give wider choice of materials and a higher potential maximum efficiency

What are the dominant factors?

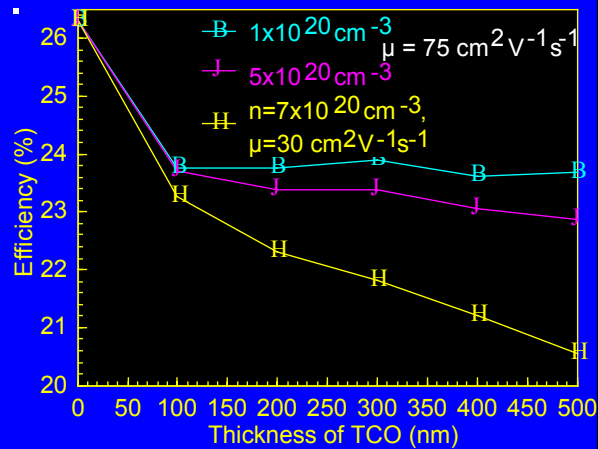
- The TCO film's optical properties are governed by Maxwell's equations
- The electron motion is described by a second order differential equation
- This was the approach used by Drude over 100 years ago

Goals of High Performance Project

- 40% laboratory demonstration of single-crystal device
 - probably monolithically grown
- 25% laboratory demonstration of thin-film polycrystalline device
 - monolithic or mechanically stacked device

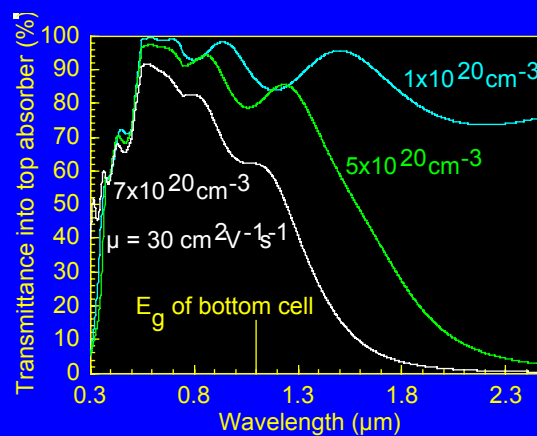
Dependence of efficiency on TCO thickness

- Always higher with lower carrier concentration
- Both high mobility materials
- Differences will increase as mobility decreases



Modeling of J_{sc}

- All TCO thicknesses are 500 nm
- $\mu = 75 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ for $n = 1$ and $5 \times 10^{20} \text{ cm}^{-3}$
- $\mu = 30 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ for $n = 7 \times 10^{20} \text{ cm}^{-3}$
- All use 50 nm CdS and 100 nm MgF_2



Everything depends on J_0 !

Wide Band Gap CuInSe₂- and CdTe- based Thin Films for Tandem Solar Cells

Bill Shafarman

Brian McCandless

Mario Gossia

Institute of Energy Conversion

University of Delaware



*Institute of Energy Conversion
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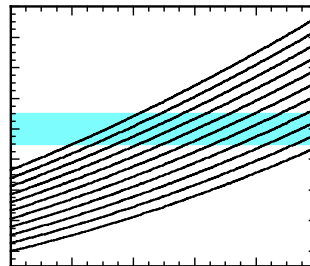
CuIn_{1-x}Ga_x(Se_{1-y}S_y)₂ Wide Band Gap Cells

CuIn_{1-x}Ga_x(Se_{1-y}S_y)₂

- Increase E_g with minimum. cation and anion alloy concentrations.
- Admittance spectroscopy lower defect concentrations than CuIn_xGa_{1-x}Se₂ or CuIn(Se_{1-y}S_y)₂ films with comparable E_g
Friedlmeier and Schock, 2nd World Conf. PVSEC, 1117 (1997).

Deposition approaches

- Se + S reaction of Cu/Ga/In precursors
 - Ga ends up at back of film
 - Requires T ≈ 600°C anneal to form uniform bandgap
Marudachalam, et. al. Appl. Phys. Lett 67, 3978 (1995).

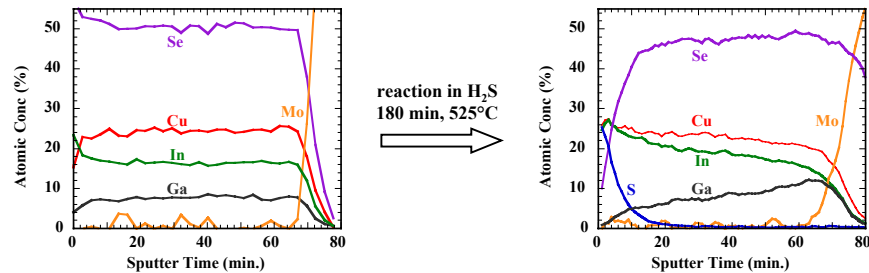


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$\text{CuIn}_{1-x}\text{Ga}_x(\text{Se}_{1-y}\text{S}_y)_2$ Wide Band Gap Cells

- S replacement reaction: $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2 + \text{H}_2\text{S} \Rightarrow \text{CuIn}_{1-x}\text{Ga}_x(\text{Se}_{1-y}\text{S}_y)_2$.
 - Films reacted in combinations of flowing $\text{H}_2\text{S} + \text{Ar} + \text{O}_2$ at 1 atm.
 - Time to uniformly incorporate S into CuInSe_2 :
 - ✧ ~ 1 hour for Cu-rich films. Engelmann, et. al., *Thin Sol. Films* 387, 14 (2001).
 - ✧ >> 8 hours for Cu-poor (device quality) films.
 - Ga moves to back of film (toward Mo contact).



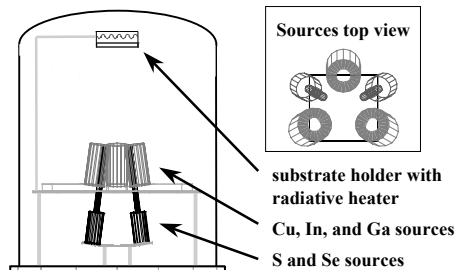
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$\text{CuIn}_{1-x}\text{Ga}_x(\text{Se}_{1-y}\text{S}_y)_2$ Wide Band Gap Cells

Elemental evaporation - critical issues.

- Thermal control of Se and S evaporation sources.
 - P_{vapor} requires $T > 100 - 300^\circ\text{C}$, while Cu, In, Ga at $1100 - 1400^\circ\text{C}$.
- Corrosivity of S vapor in deposition chamber and vacuum system.



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CdTe-based Wide Band Gap Cells

Material selection criteria:

- Complete miscibility
- Isostructural, isoelectronic p-type
- Chemical stability (e.g., HgTe unstable at $T > 400^\circ\text{C}$)
- minimize perturbation from CdTe

$\text{Cd}_{1-x}\text{Zn}_x\text{Te}$

Compound	E_g Range (eV)	End-Point Structure	Miscibility Gap ?
Cation Substitution			
$\text{Cd}_{1-x}\text{Zn}_x\text{Te}$	1.49 – 2.25	ZB - ZB	N
$\text{Hg}_{1-x}\text{Cd}_x\text{Se}$	0.10 – 1.73	ZB - W	N
$\text{Hg}_{1-x}\text{Zn}_x\text{Te}$	0.15 – 2.25	ZB - ZB	N
Anion Substitution			
$\text{CdTe}_{1-x}\text{S}_x$	1.49 – 2.42	ZB - W	Y
$\text{CdTe}_{1-x}\text{Se}_x$	1.49 – 1.73	ZB - W	N
$\text{CdSe}_{1-x}\text{S}_x$	1.73 – 2.42	W - W	N
$\text{HgTe}_{1-x}\text{S}_x$	0.15 – 2.00	ZB - ZB	?
$\text{HgSe}_{1-x}\text{S}_x$	0.10 – 2.00	ZB - ZB	N



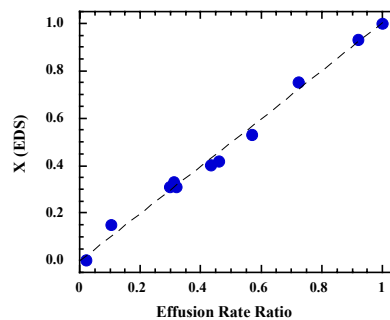
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$\text{Cd}_{1-x}\text{Zn}_x\text{Te}$ Wide Band Gap Cells

Deposition:

- Evaporation from CdTe and ZnTe compounds
 - Control film composition by relative effusion rate ratio
 $r_{\text{ZnTe}} / (r_{\text{CdTe}} + r_{\text{ZnTe}})$
 - High sticking coefficient
- $T_{\text{SS}} = 325^\circ\text{C}$
- glass/ITO/CdS substrates



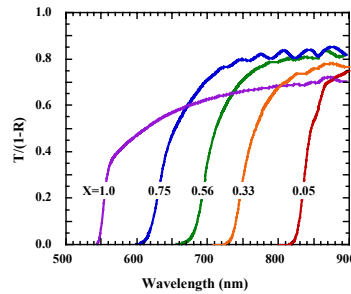
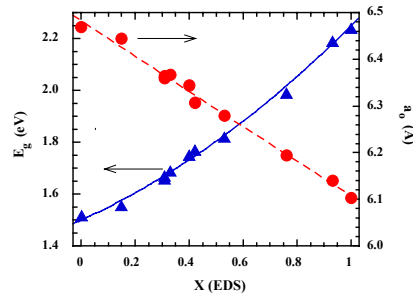
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$Cd_{1-x}Zn_xTe$ Wide Band Gap Cells

$Cd_{1-x}Zn_xTe$ films have been deposited over entire composition range.

- Optical band gap $1.5 \leq E_g \leq 2.55$ eV and bowing parameter $b = 0.3$.
- Single phase films with linear variation in lattice parameter a_0 determined from x-ray diffraction.
- Sub- E_g optical transmission = 70 - 90% over target band gap range.

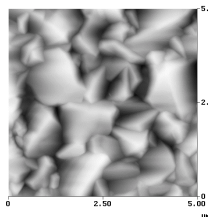


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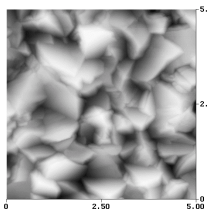
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$Cd_{1-x}Zn_xTe$ Morphology: $T_{SS} = 325^\circ C$

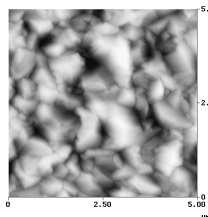
41110
 $d = 3.8 \mu m$
rms ~ 33 nm
 $x \sim 0.05$



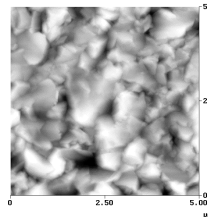
41112
 $d = 4.0 \mu m$
rms ~ 35 nm
 $x \sim 0.10$



41111
 $d = 2.8 \mu m$
rms ~ 34 nm
 $x \sim 0.30$



41113
 $d = 3.5 \mu m$
rms ~ 21 nm
 $x \sim 0.45$



- Faceted morphology
- Grain size \downarrow as $x \uparrow$
- RMS constant until $x \sim 0.45$



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Transparent back contact for $\text{CuIn}_{1-x}\text{Ga}_x\text{Se}_2$

Requirements for contact include:

- chemical stability in Se atmosphere
- ohmic contact
- adhesion and structural stability

Evaluate available TCO materials

- Calculate equilibrium stability in Se_2 at 400°C
- ZnO - actually $\text{ZnO}:\text{Al}_2\text{O}_3$ (2%)
- ITO - actually $\text{In}_2\text{O}_3:\text{SnO}_2$ (9%)

reaction, at 400°C	ΔG_{rxn} (kcal/mol)
$2\text{ZnO} + \text{Se}_2 \rightarrow 2\text{ZnSe} + \text{O}_2$	104
$2\text{Al}_2\text{O}_3 + 3\text{Se}_2 \rightarrow 2\text{Al}_2\text{Se}_3 + 3\text{O}_2$	416
$2\text{In}_2\text{O}_3 + 3\text{Se}_2 \rightarrow 2\text{In}_2\text{Se}_3 + 3\text{O}_2$	172
$\text{SnO}_2 + \text{Se}_2 \rightarrow \text{SnSe}_2 + \text{O}_2$	74



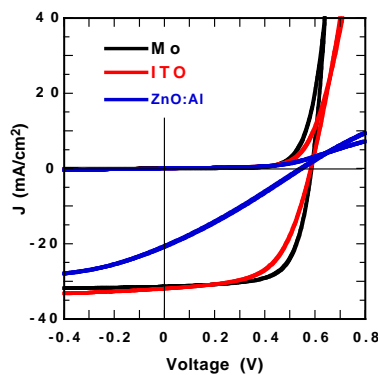
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Transparent back contact for $\text{Cu}(\text{InGa})\text{Se}_2$

Deposit $\text{Cu}(\text{InGa})\text{Se}_2$ on Mo, ITO, and ZnO

- In a single $\text{Cu}(\text{InGa})\text{Se}_2$ run
- all contacts deposited on standard soda lime glass
- $T_{\text{SS}} = 400^\circ\text{C}$
- $\text{Ga}/(\text{In}+\text{Ga}) = 0.32$, $E_g = 1.20\text{eV}$
- no apparent difference in composition, morphology



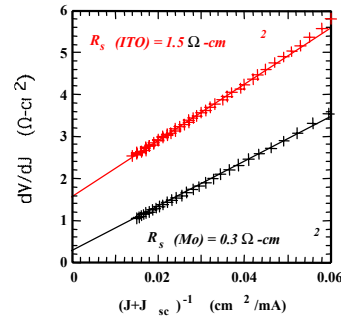
Institute of Energy Conversion
University of Delaware

High Performance PV
October 18, 2001

Transparent back contact for Cu(InGa)Se₂

contact	Voc (V)	Jsc (mA/cm ²)	FF (%)	eff (%)	Roc (Ω·cm ²)
Mo	0.587	31.4	67.7	12.5	2.0
ITO	0.584	32.0	58.4	10.9	3.7
ZnO	0.544	20.7	27.7	3.1	23

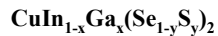
- ITO/Cu(InGa)Se₂ cell comparable to Mo/Cu(InGa)Se₂ except series resistance
- as-deposited
 - $R_{\text{sheet}}(\text{Mo}) \approx 0.2 \Omega/\square$
 - $R_{\text{sheet}}(\text{ITO}) \approx 20 \Omega/\square$



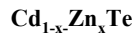
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High Performance PV
October 18, 2001

Future research focus



- Verify control of chalcogen sources. Calibrate all sources.
- Develop process to deposit films with uniform composition. Characterize films, devices.



- Determine effects of post-deposition treatments in halide, oxygen and inert ambient at 300 - 500°C.
- Fabricate and characterize devices.



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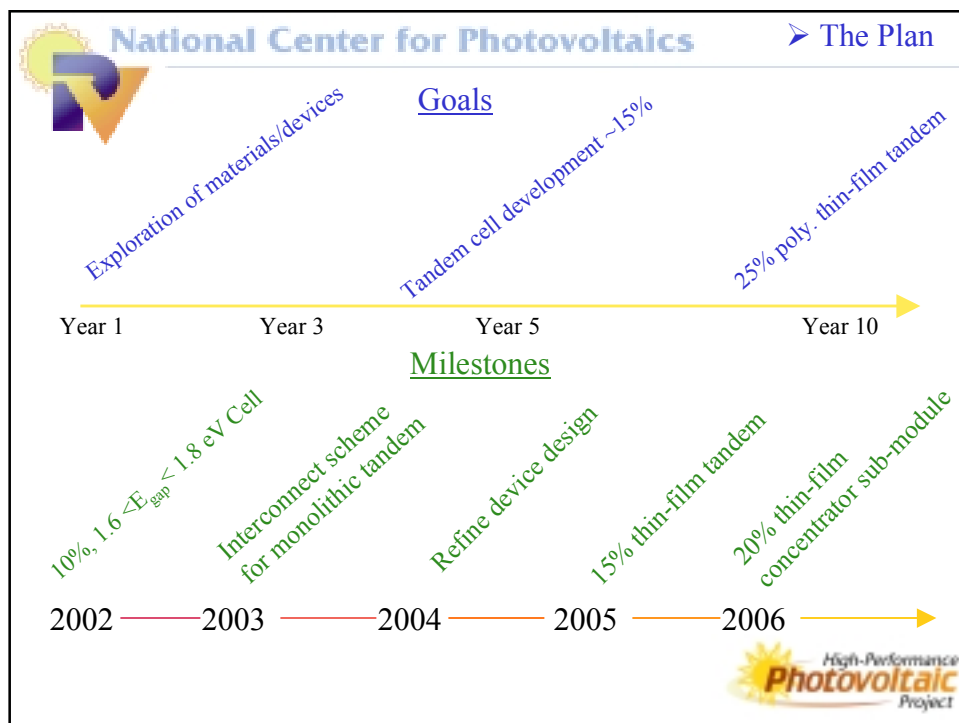
High Performance PV
October 18, 2001


National Center for Photovoltaics

NCPV Thin-Film Tandem Research

*D.L. Young, T.J. Coutts, T. Gessert, S. Ward, R. Noufi, S. Asher,
K. Emery, D. Levi, H. Moutinho, Y. Yan, P. Sheldon, M. Symko-Davies*







Address at Rice University on the Nation's Space Effort
President John F. Kennedy
Houston, Texas
September 12, 1962

"We set sail on this new sea because there is new knowledge to be gained, and new rights to be won, and they must be won and used for the progress of all people."

"We choose to go to the moon. We choose to go to the moon in this decade and do the other things, not because they are easy, but because they are hard, because that goal will serve to organize and measure the best of our energies and skills, because that challenge is one that we are willing to accept, one we are unwilling to postpone, and one which we intend to win, and the others, too."

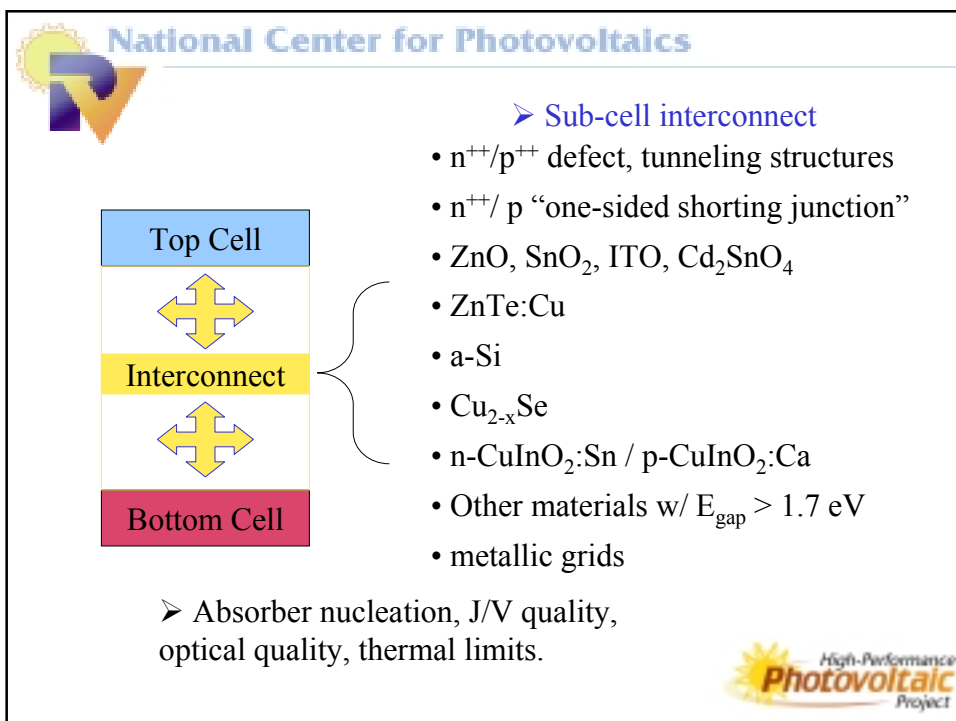
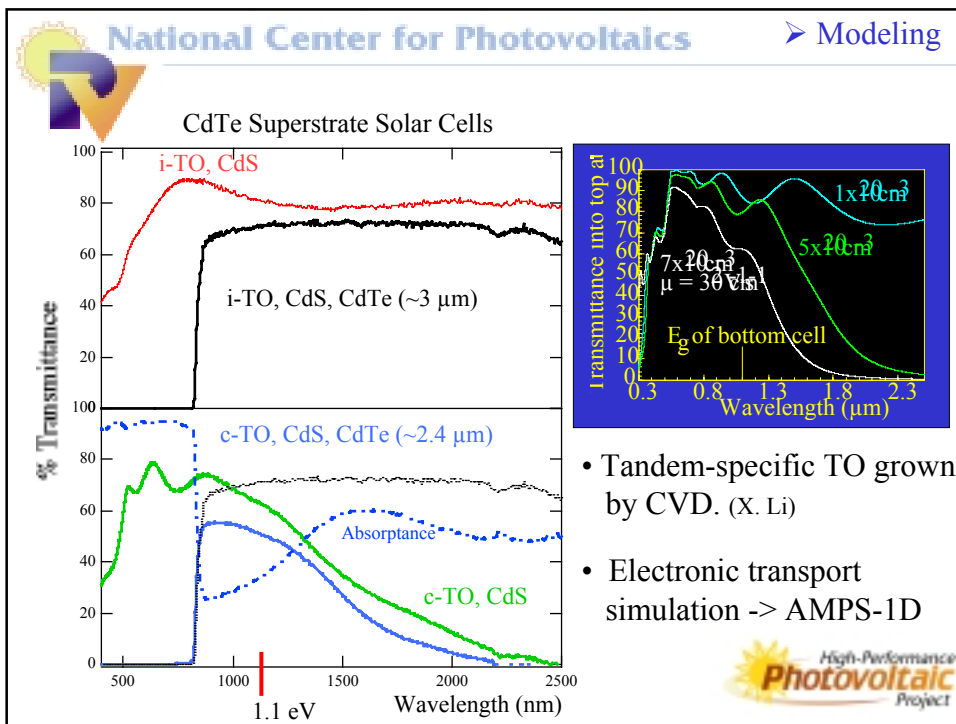
"But if I were to say, my fellow citizens, that we shall send to the moon, 240,000 miles away ...a giant rocket more than 300 feet tall,... **made of new metal alloys, some of which have not yet been invented**, capable of standing heat and stresses several times more than have ever been experienced, fitted together with a precision better than the finest watch, carrying all the equipment needed for propulsion, guidance, control, communications, food and survival, on an untried mission, to an unknown celestial body, and then return it safely to earth,... and do all this, and do it right, and do it first before this decade is out--**then we must be bold.**"

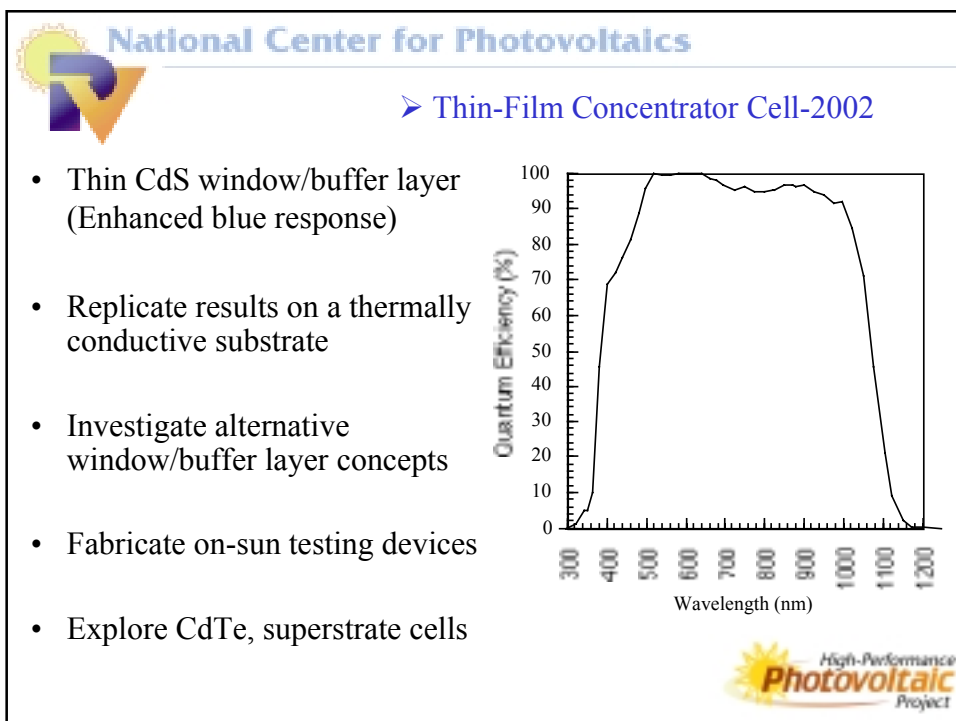
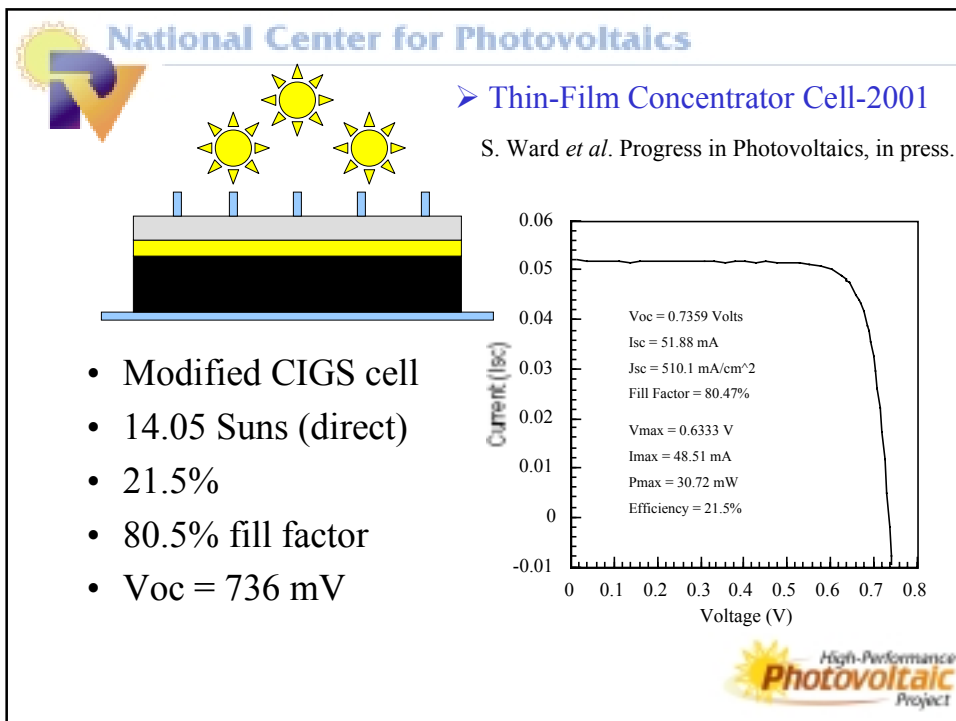


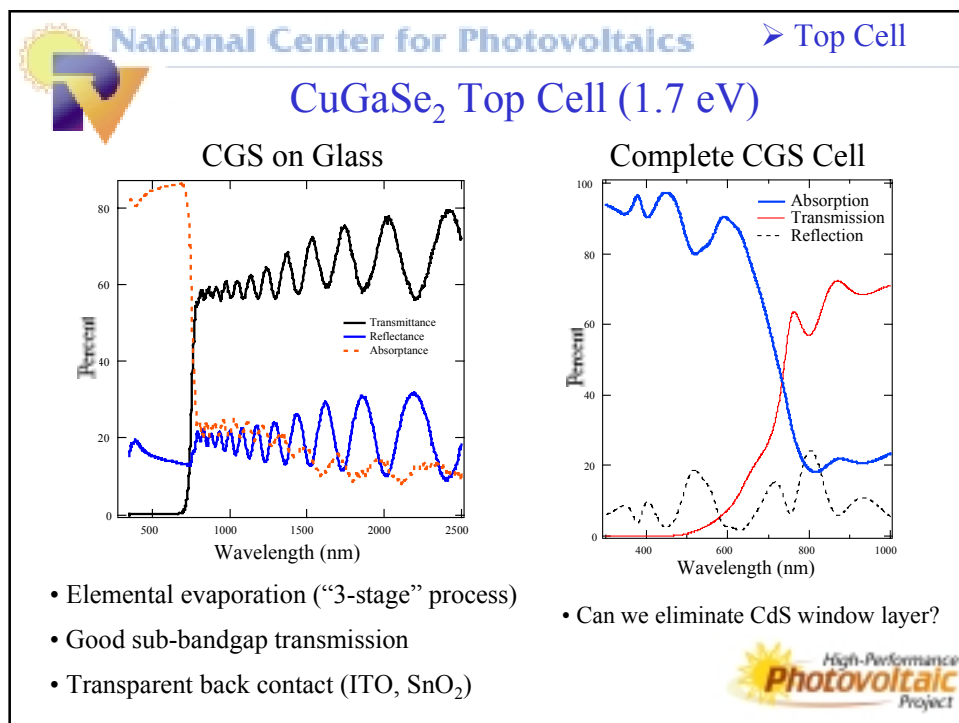
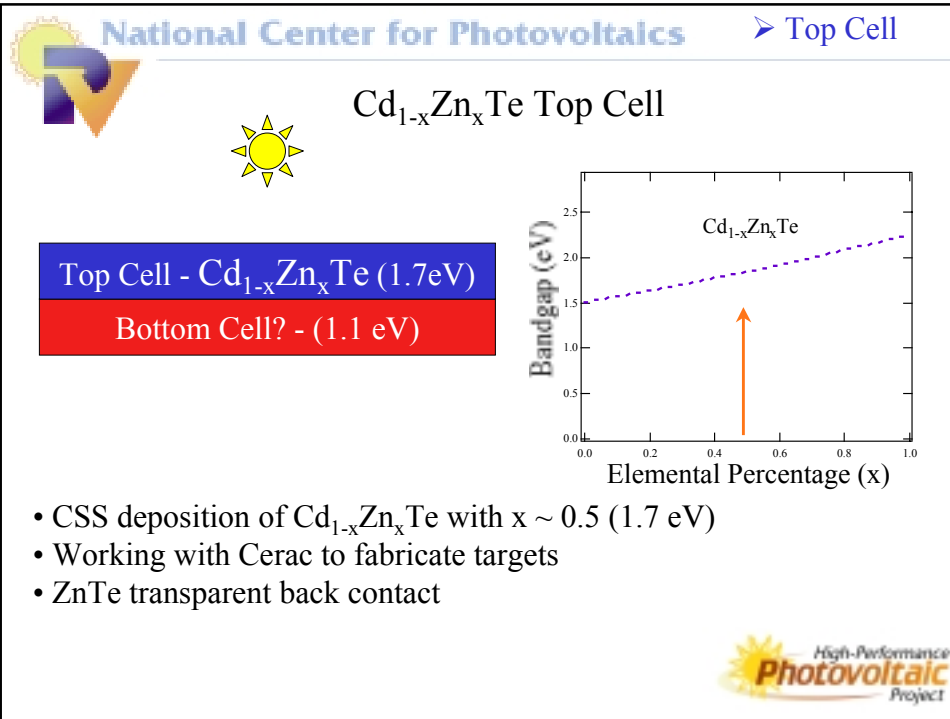
NCPV Thin-Film Tandem Research - 2002 A.O.P.

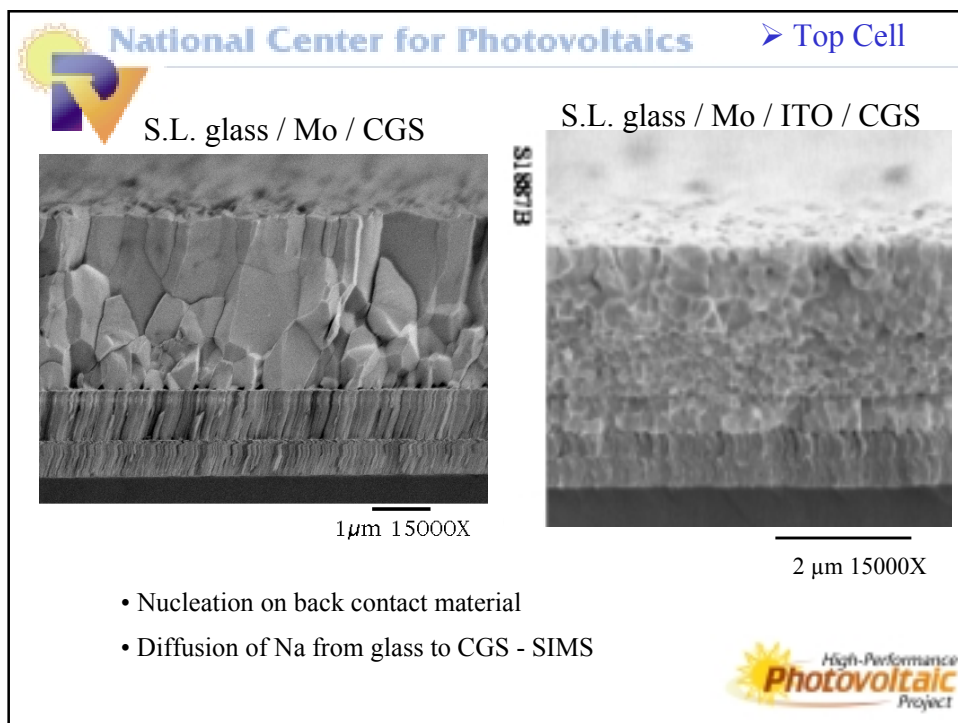
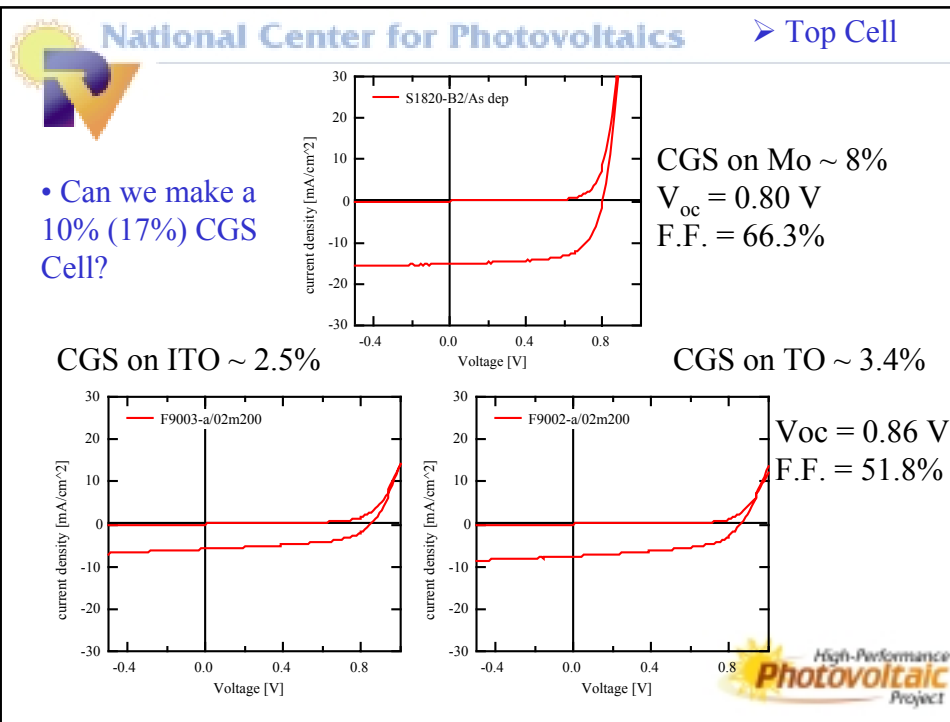
- Modeling
- Thin film concentrator cells
- Sub-cell interconnect
- Exploration of wide bandgap top cell materials
- New equipment and personnel

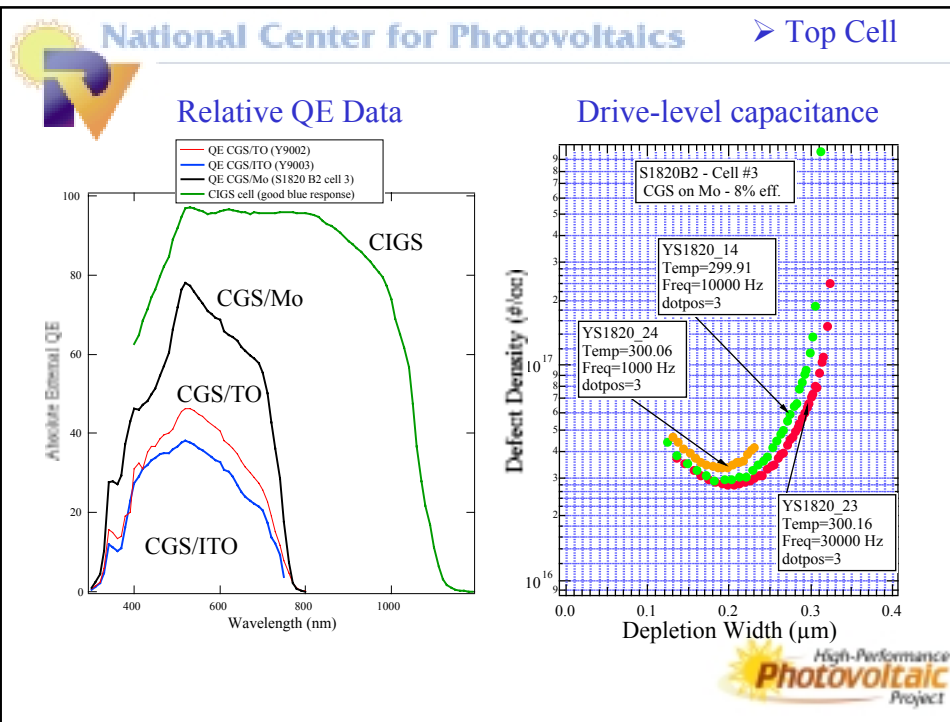












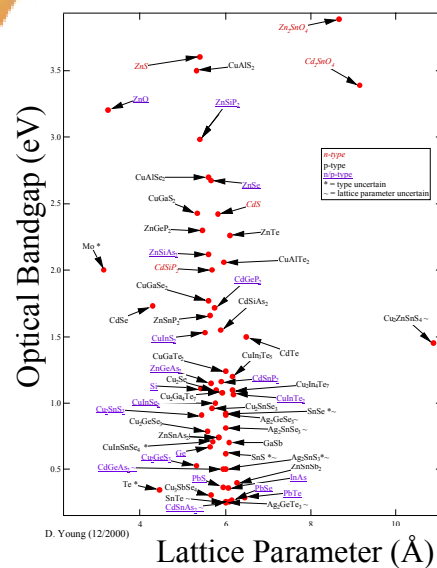
National Center for Photovoltaics ➤ Equipment and Personnel

- Retool existing Solar-X evaporator
 - 5 sources
 - Active substrate cooling
- Recruit in-house specialist to work on High Performance goals.
 - ~1/3 of NREL work ➡ Future PV

High-Performance Photovoltaic Project



- J.F. Kennedy 1962



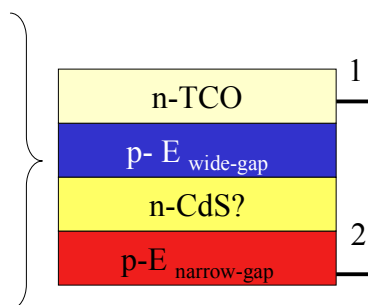
- Literature search based on optical modeling





New Materials for Tandem Cell

ZnSiAs₂ - 2.12 eV
CuGaSe₂ - 1.7
CuAlTe₂ - 2.06
ZnTe - 2.12
ZnSnP₂ - 1.66
CuAlSnSe₄ - 1.90
ZnGeP₂ - 2.3
ZnIn₂Se₄ - 2.05
ZnIn₂Te₄ - 1.87
CdGeP₂ - 1.72
CdSiP₂ - 2.05
Se (:I, VII) - 1.8



CuInSe₂ - 0.95 eV
CuInSnSe₄ - 0.71
CuGaTe₂ - 1.24
Cu₂Ga₄Te₇ - 1.08
CuInTe₂ - 1.1
ZnGeAs₂ - 1.15
Cu₂GeSe₃ - 0.79 - 0.94
Cu₂SnS₃ - 0.91
Cu₂SnTe₃ - 0.96
GaSb - 0.70
Si - 1.1



High Performance PV Research at U. of Toledo: “Polycrystalline Thin-Film Tandem PV Cells”

- First Solar, LLC, is a major lower-tier subcontractor
- University of Toledo focus:

First Year --

- magnetron sputtering of wide gap II-VI alloys for top cells with emphasis on CdZnTe
- magnetron sputtering of ZnTe:N and ZnO:Al for top cell transparent back contacts and recombination junctions
- (with FS) study of HRT layers for reduced CdS thickness
- (with FS) fabrication of four-terminal stacked device with CIS bottom cell

Second year--

- optimization of CdZnTe or other II-VI with special focus on post-deposition chloride heat treatment
- optimization of recombination junctions between wide-gap II-VI and CIS
- exploratory work on magnetron sputtered HgCdTe for bottom cells with possible fabrication of two-terminal structures with MS and/or VTD top cells
- characterization of materials-- PL, EL, Raman, XRD, Hall, photo-Hall, SEM/EDS
- device characterization including QEs of component cells of two-terminal structures

Absorber Layer, $\text{Cd}_{1-x}\text{Zn}_x\text{Te}$

Top Cell Absorber Layer Band Gap $\sim 1.7\text{-}1.85\text{ eV}$

CdTe Band Gap $\sim 1.5\text{ eV}$

ZnTe Band Gap $\sim 2.2\text{ eV}$

Preparation of $\text{Cd}_{1-x}\text{Zn}_x\text{Te}$ Films: By Magnetron Sputtering

Target: in house target fabrication with mixture & sintered $\text{Cd}_{1-x}\text{Zn}_x\text{Te}$

Challenge: Post deposition treatment

Transparent Recombination Contact

- ZnTe:N -**
- Band Gap $\sim 2.2\text{ eV}$
 - Prepared by Reactive Sputtering Technique
 - N_2 Composition: 0 - 5% in Ar

Effect of N doping

Structural: orientation changes from (111) to (220)

Morphology: grain size decreases from 130 to 36 nm

Energy Band Gap: No trend, between 2.16 to 2.22 eV

Properties of ZnTe:N Films (3% N₂)

Resistivity: ~ 8 ohm-cm (as deposited)

: ~ 3 ohm-cm (heat treated in air)

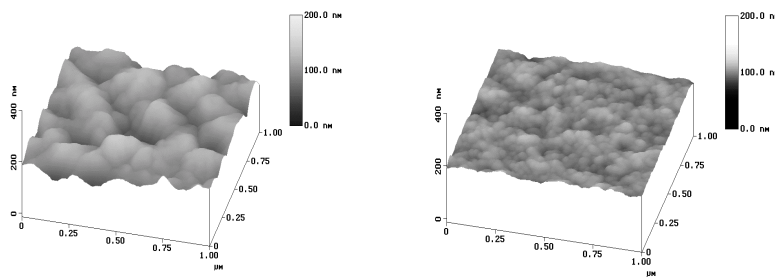
Carrier concentration: $\sim 5 \times 10^{18} \text{ cm}^{-3}$

Mobility: $\sim 0.3 \text{ cm}^2/\text{V-s}$

ZnO:Al Films: By Magnetron Sputtering Technique

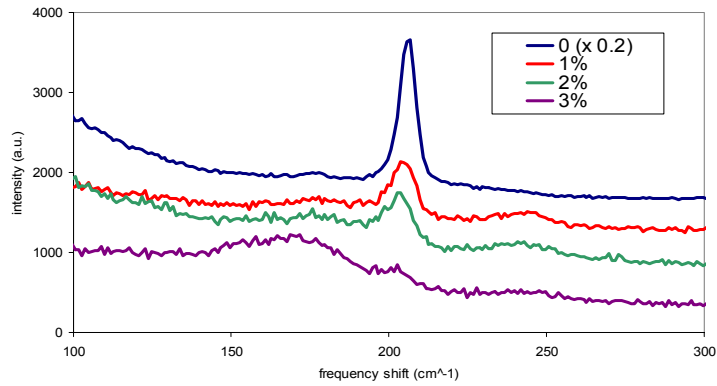
Window Layer: Use of High-resistive Tin Oxide to decrease CdS thickness

Atomic Force Microscopy of ZnTe Films



AFM images of ZnTe (left) and ZnTe:N (right) films sputtered with 3% N₂/(Ar+N₂) on microscope slides.

Raman Spectra of ZnTe and ZnTe:N films

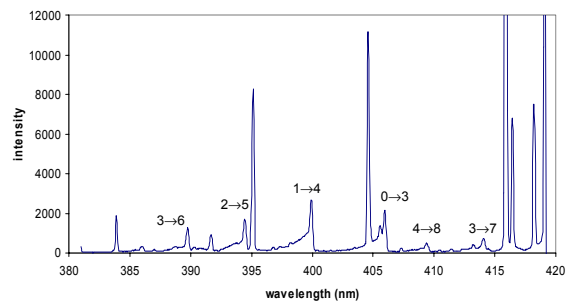


(1% and 3% spectra are offset by 1000 and 500 respectively)

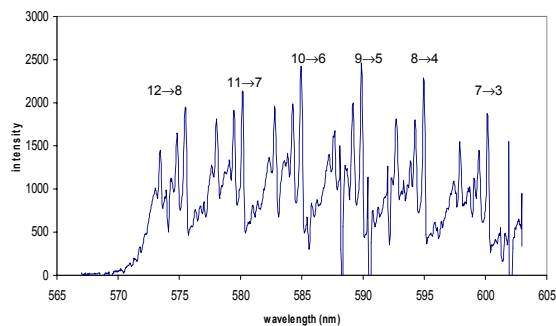
- As doping increases, the LO peak at 203 cm^{-1} decreases in intensity.
- Free carrier screening shifts LO to the TO position at 170 cm^{-1}
- Broad peak for 3% suggests density-of-states scattering appropriate for a partial amorphous film.

OES spectra of N_2 near target

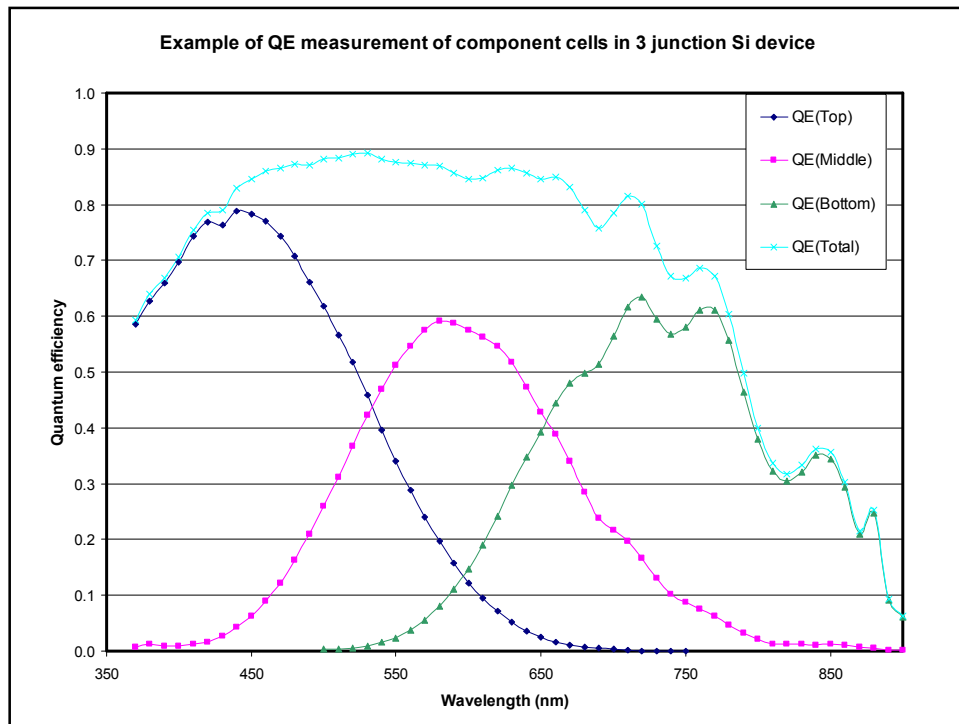
Second positive band
($\text{C}^3\Pi_u - \text{B}^3\Pi_g$)



First positive band
($\text{B}^3\Pi_g - \text{A}^3\Sigma_u^+$)



[sputter power 40W, press = 18mT. Ar lines have been subtracted.]





Plans for Global Solar Energy's High-Performance Program

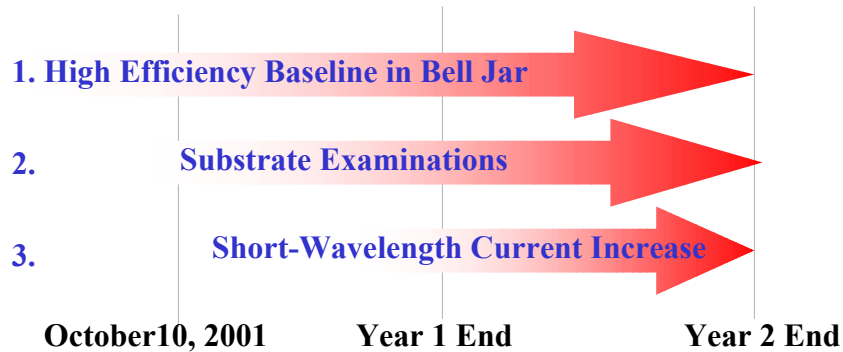
Prime Contractor: Global Solar Energy
Lower-Tier Subcontractor: ITN Energy Systems
Track: Thin-Film Concentrator



Phase I: How do we move thin film CIGS devices on metal foil toward 20% efficiency?

- 1. Eliminate efficiency differences between devices on steel and on glass.**
- 2. Increase short-wavelength current collection.**

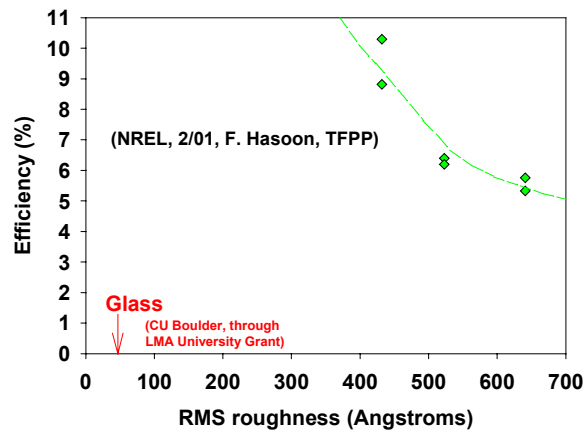
Scheduling:



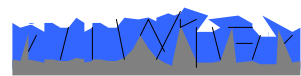
Substrate Examinations: Eliminate efficiency differences between best devices on steel and best devices on glass.

1. Substrate roughness
2. Harmful impurity diffusion, with and without barriers
3. Modified NaF incorporation
4. Substrate Temperature

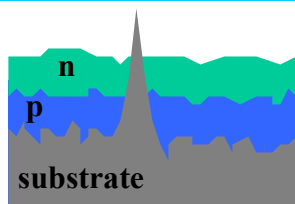
Substrate Roughness:



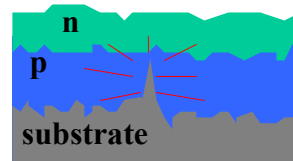
Mechanism:



Altered Grain Growth?



Shunt Paths?



Source of Impurities?

Activities:

AFM

Electropolishing

IV Characterization

Device Fabrication

Experiments:

η vs. roughness. Can we approach smoothness of glass with electropolishing?

Is roughness equally harmful for foils without harmful impurities? (Mo, Ti)

How much does Mo mask the bare substrate roughness?

Does NaF precursor morphology affect growth surface? Is it better deposited after In_2Se_3 growth?

Substrates, Harmful Impurity Diffusion

Activities:

Device Fabrication

Diffusion Barrier
Deposition

IV Characterization

SIMS

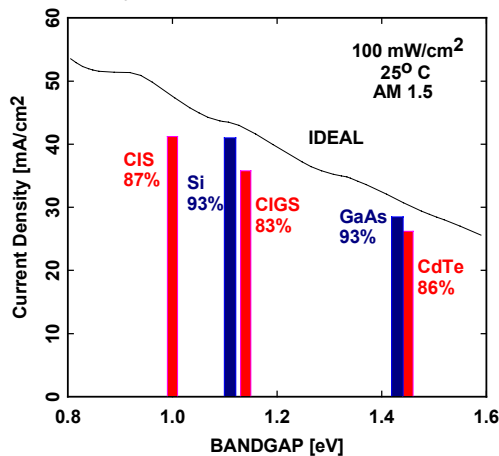
Experiments:

Device performance and Fe, Cr, C concentrations vs. Mo thickness

Device performance and Fe, Cr, C concentrations vs. barrier (such as Cr) thickness and properties

Do roughness studies on Mo and Ti foils help quantify effect of impurities?

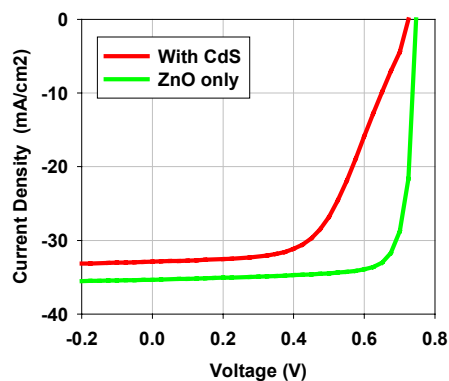
2. Why Focus On Short-Wavelength Current Increase?



- CIGS has realized only 83% of its potential current collection, compared to over 90-95% for voltage and ff.
- NREL 15% Cd-free device collects 1.7 mA/cm² more current below CdS bandgap than record 18.8% device.
- (Graph not updated to include recent CdTe record.)

Hypothesized functions of CdS:

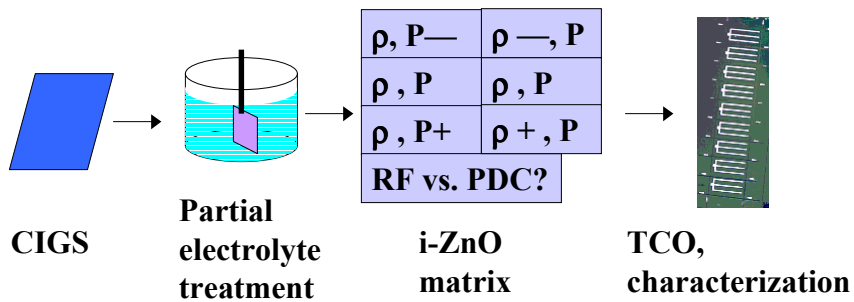
- Diffusion of Cd into CIGS
- Protection from PVD
- Creation of lattice-matched interface
- Mitigation of localized shunts
- CIGS surface reduction in bath



ADEPT modeling: CdS NOT necessary for successful band structure.

Examine each role, determine importance and how might be replaced with higher-transmission option:

◆ Mitigate localized shunts ◆ Protect from PVD



◆ Diffusion of Cd into CIGS

Activities:

Variables:

Partial Electrolyte Treatment	Immersion time	CdSO ₄ concentration
Device Fabrication	Bath temperature	NH ₄ OH concentration
IV Characterization	Applied voltage	
Comparative SIMS?	CIGS near-surface composition	

- ◆ Creation of lattice-matched interface
- ◆ CIGS surface reduction in bath

Deposit very thin CdS layers?

- Slow down bath with lower thiourea concentration or lower temperature.
- Vary thickness by varying sample immersion time.

Valid examination *if*

- Oxidation creates active defects at CIGS/CdS interface, but not at CdS/ZnO interface.
- Lattice matching-related defects between CdS and ZnO are less active than those between CIGS and ZnO.
- Or, junction is entirely in CIGS.

Otherwise, defects are still very close to junction.

SUMMARY

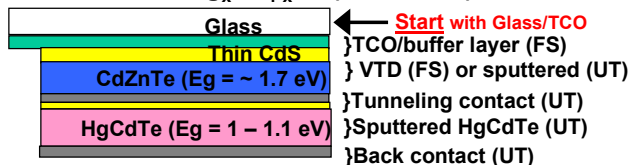
Program objectives are to:

- 1. Determine the important variables in the efficiency differences between the best devices on glass and the best devices on steel.**
- 2. Improve short-wavelength current collection by determining the important functions of the CdS and replacing it.**
- 3. Demonstrate at least 16% efficient CIGS devices on metal foil that demonstrate improved blue light collection and insignificant efficiency degradation between glass and steel.**

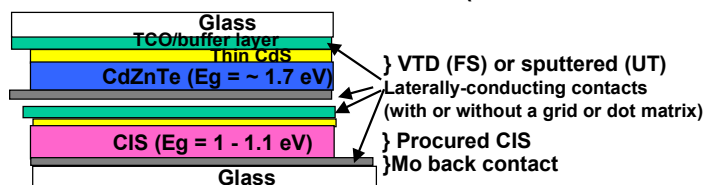


Tandem device structures under investigation

- **Superstrate** CdS/CdZnTe, $\text{Hg}_x\text{Cd}_{1-x}\text{Te}$ (2 terminal):



- **Superstrate** CdS/CdZnTe in 4-terminal device (with CIS bottom cell):



- **Substrate** CdS/CdZnTe grown on CIS or HgCdTe bottom cell (2 terminal) (UT)

10/18/01, High Performance PV Kickoff Meeting, First Solar, LLC



Task 1: Make and supply substrates to UT

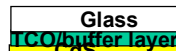
- Supply:

- glass/TCO/HRT:



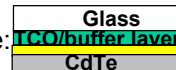
- CdS/CdTe sputtered top cell

- glass/TCO/HRT/CdS:



- CdTe sputtering

- glass/TCO/HRT/CdS/CdTe:



- Tunneling contact
- Sputtered bottom cell
- Mechanical-stack

- **Comments:**

- UT has found that some FS CdCl_2 -treated glass/TCO/HRT/CdS/CdTe substrates are more suited than others to a no-etch back contact process. We have worked to identify substrates that will perform well with their process.

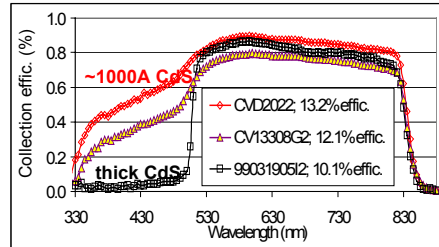
10/18/01, High Performance PV Kickoff Meeting, First Solar, LLC



Task 2: Investigate tailoring buffer/thin CdS

- Buffer layer and thin CdS with optimized properties are important to high efficiency with single junction CdS/CdTe cells: →

“” even more important in tandem cells.



- Capabilities to investigate buffer layers include an atmospheric-pressure chemical vapor deposition (APCVD) system with 60cm-web and 1 plate/min throughput →



APCVD system for buffer layer deposition

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Task 3: Explore feasibility of CdZnTe by VTD

- Task: Explore the feasibility of depositing CdZnTe with controlled levels of Zn with research-sized Vapor Transport Deposition (VTD) system; characterize films by spectrophotometry, EDS and SEM.

Small area
VTD system
(10 cm x 10
cm samples):



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Other year 1 tasks for First Solar

- Interconnect and laminate top cells with Cl(G)S bottom cells for mechanically stacked cells
- Light-soak and characterize single and dual-junction cells developed as part of this effort

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Year 2 tasks for First Solar

- Make and supply samples to UT to support magnetron-sputtered top-cell absorber layer, top-cell-contact, bottom-cell, and mechanical-stacked cell development efforts:
 - glass/TCO/HRT; glass/TCO/HRT/CdS; glass/TCO/HRT/CdS/CdTe
- Optimize HRT/thin-CdS layers for use in top cells with CdZnTe
- Deposit by VTD CdZnTe with controlled levels of Zn for use with two-terminal and four-terminal tandem cells; characterize films by spectrophotometry, EDS and SEM.
- Interconnect and laminate top cells with Cl(G)S bottom cells for mechanically stacked cells
- Light-soak and characterize single and dual-junction cells developed as part of this effort

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Some other relevant facilities

- **Processing at Technology Center:**
 - Module scale back-contact deposition and processing equipment
 - Sputter-deposition and e-beam evaporation systems
 - Ventilation hoods and wet-chemistry for pre- or post-contact treatments
 - 2 ovens for heat treatments
 - Small-area (100 cm²) APCVD system for specialty buffer layers
 - Small-sample encapsulation
- **Finishing line at factory** for 60 cm x 120 cm modules (1 plate/min designed capacity)
- **Accelerated-life testing (ALT) systems:**
 - Light-soak stations for 400 10cm x 10cm samples and 26 modules
 - 3 ovens for dark stress ALT (with N₂ purge and external bias stress options)
- **Characterization systems:**
 - Xenon-lamp solar simulator (measures up to 18 cells at once; tracked in database)
 - CV/CF system for carrier-concentration and trap characterization
 - Varian Cary 500 spectrophotometer with integrating sphere
 - SEM with EDS capability
 - Solar cell spectral response measurement system (monochromator-based with Xenon lamp source)
 - X-Y mapping systems with 4-pt resistance, CdS and CdTe thickness by transmission, SPV, and PL of CdTe.

10/18/01, High Performance PV Kickoff Meeting, First Solar, LLC

Development of a II-VI-Based High Performance, High Band Gap Device for Thin-Film Tandem Solar Cells

C. S. Ferekides and D. L. Morel
ELECTRICAL ENGINEERING DEPARTMENT
CENTER FOR CLEAN ENERGY AND VEHICLES
UNIVERSITY OF SOUTH FLORIDA
TAMPA, FL

SUPPORTED BY THE NATIONAL RENEWABLE ENERGY LABORATORY

Department of Electrical Engineering, Center for Clean Energy and Vehicles,
Presented at the 1st High Performance PV Meeting, NREL, October 18, 2001.



OUTLINE

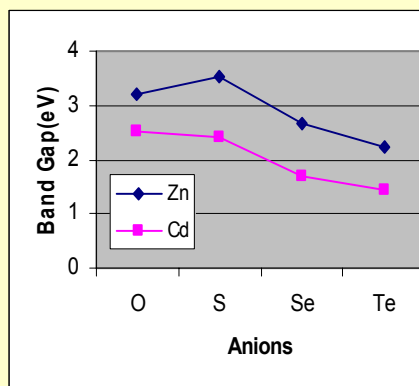
- **Tandem solar cells**
 - Why? to reach 25% efficiencies
 - How?
 - Using a 16-18% high band gap top cells
 - Top $E_g=1.7$ eV, bottom $E_g=1.0$ eV (CIGS)
 - II-VI compounds can be used for top cell
- **Device structure**
- **AMPS Modeling**
 - Bottom cell vs. thickness of top absorber
 - Light J-V
 - SR
 - J_{sc} and η
 - Top Cell performance vs. thickness
 - Tandem Cells
- **Approach**

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Presented at the 1st High Performance PV Meeting, NREL, October 18, 2001.



Background

- The band gap combination 1.7/1.0 eV is near-ideal for a 2-terminal tandem
- CIGS with a band gap of 1.0 eV and an efficiency of 15% is the clear choice for the bottom cell
- To reach 25% the top cell will have to be in the 16-18% range since it will contribute about 2/3rds of the output
- As seen in figure 1, II-VI compounds have the requisite band gaps
- $\text{Cd}_{1-x}\text{Zn}_x\text{Te}$ (CZT) covers the range 1.45-2.2 eV and can capitalize on successes with CdTe technology
- CdSe has a band gap near the ideal of 1.7 eV and is a binary

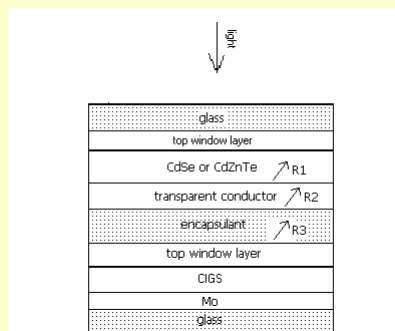


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Device Structure

- A 4-terminal structure is proposed as shown in figure 2.
- Top Cell: II-VI; Bottom Cell: CIGS
- Preliminary calculations indicate that reflection losses to the bottom cell can be kept to 10%
- Key issue: Transparent Contact for top cell.
- Window Layers – several options
- Substrates: Glass!

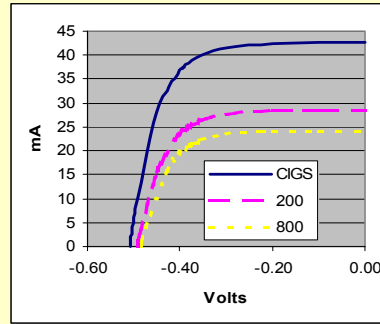


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AMPS Simulation – Bottom Cell vs. Thickness of Top Absorber

- Achieved parameters for a 14.7% 1.0 eV CIGS cell are used for the bottom cell ($J_{sc} = 42.6 \text{ mA/cm}^2$, $V_{oc} = 508 \text{ mV}$, $FF = 0.68$.)
- The thickness of the top 1.7 eV cell is varied from 200 to 800 nm.

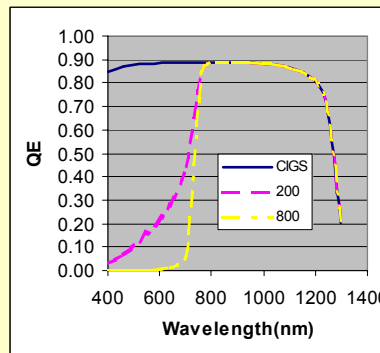


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AMPS Simulation – Bottom Cell QE vs. Thickness of Top Absorber

- There is a large drop in J_{sc} for 200 nm, but the drop slows significantly as the top cell thickness is increased to 800 nm

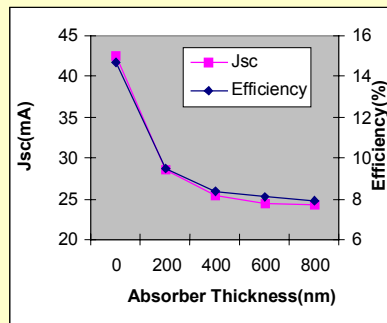


Department of Electrical Engineering, Center for Clean Energy and Vehicles,
Presented at the 1st High Performance PV Meeting, NREL, October 18, 2001.



AMPS Simulation – Bottom Cell Performance

- J_{sc} asymptotes toward 25 mA/cm²
- Efficiency of the bottom cell is in the range of 8-9% for the 400-800 nm thickness range

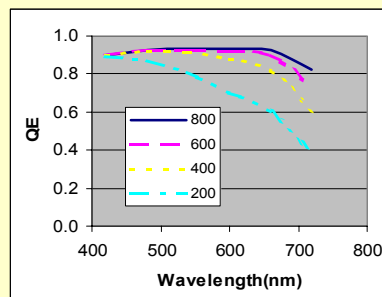


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AMPS Simulation – Top Cell OE

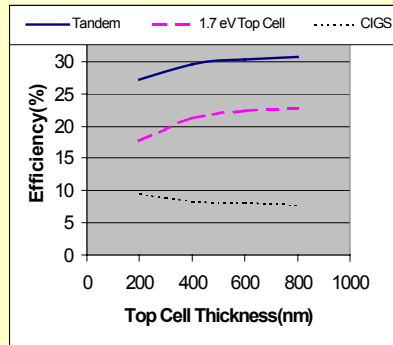
- Not much loss in the top cell occurs when its thickness is reduced to the 400-600 nm range



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AMPS Simulation – Tandem Performance (ideal)

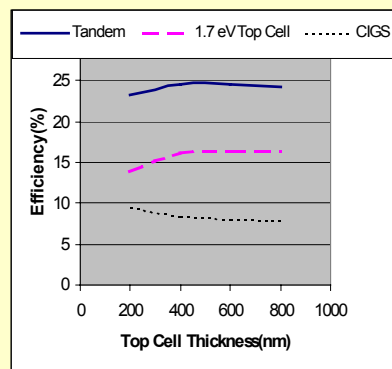


- Using idealized values for a top cell of E_g 1.7 eV, the efficiency is in the low 20% range for 400-800 nm thickness
- This results in a tandem efficiency of 30+%

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AMPS Simulation – Tandem Performance (real!)



- Using typical values of parameters (CdTe) yields a top cell efficiency in the desired range above 16% and a tandem efficiency of 25%.

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Deposition Technology – Current Status

➤ **Deposition Technology:**

- New deposition system with co-evaporation capabilities will be constructed for this project
- Close spaced sublimation (available)
- RF-sputtering

➤ **Challenges: develop ABSORBER and its TRANSPARENT CONTACT**

➤ **Project started on 09/01:**

- Begun deposition of II-VI films using existing deposition capabilities.

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Presented at the 1st High Performance PV Meeting, NREL, October 18, 2001.





NREL High Performance PV Program



Identification of Critical Paths in the Manufacture of Low-Cost High-Efficiency CGS/CIS Two-Junction Tandem Cells

by

Oscar D. Crisalle
University of Florida
Gainesville, FL

Program Kick-Off Meeting, Golden, CO, October 18, 2001



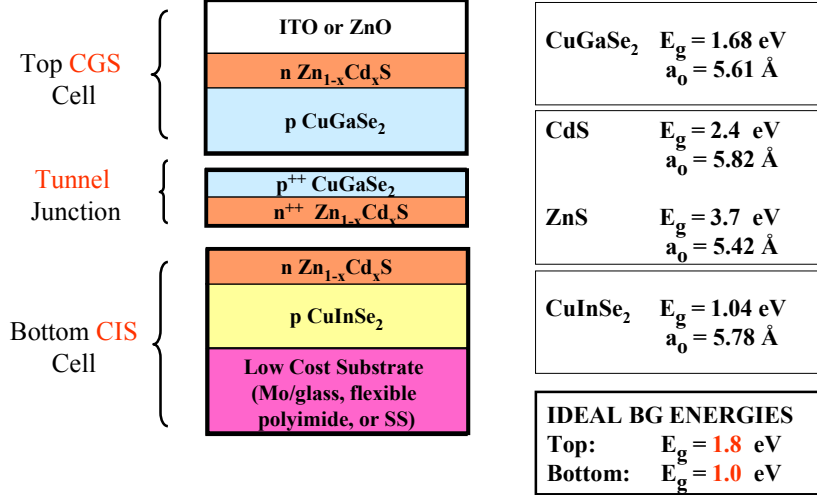
Principal Investigators



- Prof. Oscar D. Crisalle
 - Chemical Engineering Department
 - Instrumentation, process modeling and control
- Prof. Sheng S. Li
 - Electrical & Computer Engineering Department
 - Device design, characterization, & optimization
- Prof. Tim J. Anderson
 - Chemical Engineering Department
 - Materials growth, characterization, & optimization



Proposed Tandem Cell Structure



Critical Issues



- **Tunnel junction:**
 - Fabrication of a transparent, low resistance, lattice-matched, and heavily-doped junction.
 - Low temperature deposition process
- **Top cell (CGS)**
 - Improved conversion efficiency
 - Low temperature deposition process



Approach



- Study each component separately during Phase I
- Reduce complexity by growing single-crystal films on GaAs (use low-cost substrates in Phase II)

ZnO : Al	ZnO : Al	Metal contact
n Zn _{1-x} Cd _x S	n Zn _{1-x} Cd _x S	p ⁺⁺ CuGaSe ₂
p CuGaSe ₂ (a ₀ = 5.61 Å)	p CuInSe ₂ (a ₀ = 5.78 Å)	n ⁺⁺ Zn _{1-x} Cd _x S
p ⁺ GaAs (a ₀ = 5.653 Å) Single-Crystal Substrate	Graded Layer: p Cu(In,Ga)Se ₂	n ⁺ GaAs
	p ⁺ GaAs (a ₀ = 5.653 Å) Single-Crystal Substrate	Semi-Insulating GaAs Substrate

(a) Top-Cell Structure (CGS)

(b) Bottom- Cell Structure (CIS)

(c) Tunnel-Junction Structure



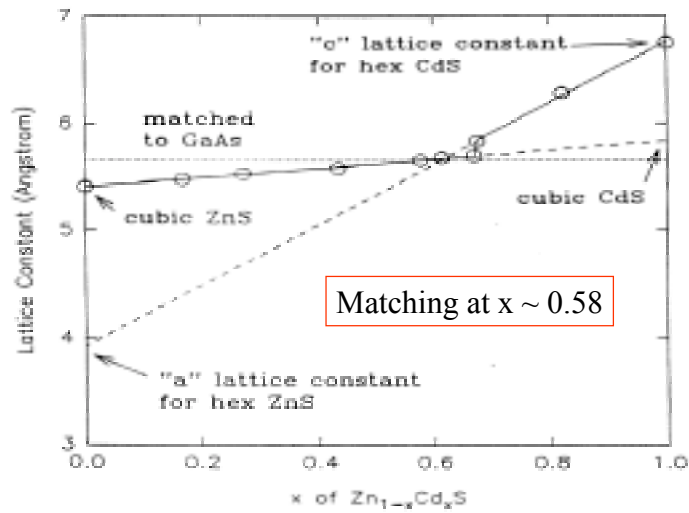
Project Goals



- Identify the critical manufacturing and materials issues for achieving a **25% conversion efficiency CGS/CIS** two-junction tandem solar cell.
- Fabricate a **bottom CIS cell** with a conversion efficiency in the **13% - 15 %** range.
- Fabricate a **top CGS cell** with a conversion efficiency in the **10% - 13 %** range.
- Fabricate a low resistance and high quality **tunnel junction** interconnecting the top and bottom cells.



$\text{Zn}_{1-x}\text{Cd}_x\text{S}$ Lattice



Migration-Enhanced Epitaxial Growth



MEE
Reactor





Bottom CIS Cell - Activities Proposed



ZnO:Al
n $\text{Zn}_{1-x}\text{Cd}_x\text{S}$
p CuInSe ₂
Graded p Cu(In,Ga)Se ₂
p ⁺ GaAs Single Crystal Substrate

- Take advantage of standard processing technology
- Grow and characterize single-crystal epitaxial CIS by Migration Enhanced Epitaxy (MEE)
- Single-crystal n⁺ GaAs substrates plus graded p Cu(In,Ga)Se₂ (or p In_xGa_{1-x}As) to reduce the dislocation density and grade the band gap energy
- Grow and characterize Zn_{1-x}Cd_xS layer
- Deposit ZnO:Al layer via sputtering.



... Bottom CIS Cell Activities Proposed



ZnO:Al
n $\text{Zn}_{1-x}\text{Cd}_x\text{S}$
p CuInSe ₂
Graded p Cu(In,Ga)Se ₂
p ⁺ GaAs Single Crystal Substrate

- Dark- and photo- current-voltage (I-V) and capacitance-voltage (C-V) characteristics and spectral response measurements to determine the quantum efficiency.
- Develop an optimal device model for the bottom CIS cell and identify the physical parameters and optimal thickness of each layer using AMPS-ID device simulation program.



Top CGS Cell - Activities Proposed



ZnO:Al
n $\text{Zn}_{1-x}\text{Cd}_x\text{S}$
p CuGaSe ₂
p ⁺ GaAs Single Crystal Substrate

- Grow single-crystal and epitaxial CGS on single crystal substrates by MEE and characterize the CGS absorber layers.
- Investigate an electrodeposition process for CGS as (low temperature process and low cost alternative)
- Optimize the electronic properties of the CGS absorber layer (net hole concentration $\sim 10^{16} \text{ cm}^{-3}$, mobility, and lifetime).
- Seek to increase the conversion efficiency (consider the addition of S).



... Top CGS Cell - Activities Proposed



ZnO:Al
n $\text{Zn}_{1-x}\text{Cd}_x\text{S}$
p CuGaSe ₂
p ⁺ GaAs Single Crystal Substrate

- Investigate temperature, composition and other factors affecting the deposition of the $\text{Zn}_{1-x}\text{Cd}_x\text{S}$ buffer layer
 - Consider CBD and MOCVD processes
- Perform the dark- and photo- I-V characteristics and spectral response measurement of the top CGS cell.
- Develop optimal device model for the top CGS cell and identify the physical parameters & optimal thickness of each layer using AMPS-ID



Tunnel Junction - Activities Proposed



Metal contact
p^{++} CuGaSe ₂
n^{++} Zn _{1-x} Cd _x S
n^{+} GaAs
Semi-Insulating GaAs Substrate

- Construct a prototype tunnel junction diode
 - Determine if CGS can be doped p-type in the 10^{18} cm^{-3} range
 - Investigate venues for n^{++} Zn_{1-x}Cd_xS layer deposition
 - Explore alternatives for n^{++} doping of the Zn_{1-x}Cd_xS layer
 - ▼ Utilize intrinsic defects in CBD processing
 - ▼ Utilize extrinsic dopants (group III (Ga or In) or halide atoms (Cl, Br, or I)).



... Tunnel Junction - Activities Proposed



Metal contact
p^{++} CuGaSe ₂
n^{++} Zn _{1-x} Cd _x S
n^{+} GaAs
Semi-Insulating GaAs Substrate

- Characterize the effect of processing via I-V characteristics and the peak current density under different annealing temperatures and duration
- Characterize the n^{++} Zn_{1-x}Cd_xS layer by XRD, XPS, SEM, and Hall and transmission measurements
- Develop a tunnel junction diode model and identify the physical parameters and optimal thickness of the tunnel junction layer using AMPS-ID.



Milestones- Phase I Y1



- Produce epitaxial CIS and CGS on lattice-matched substrates.
- Characterize electrical and optical properties of the tunnel junction; determine the optimal thickness/doping level for low optical & electrical losses.
- Develop device model for each component of tandem cell using AMPS-1D device simulation program and predict single-junction cell performance.
- Develop and evaluate a one-step electrodeposition method for growing CGS films.



Milestones- Phase I Y2



- Identify growth parameters for a 10% - 13% efficiency CGS cell
- Identify growth parameters for a 13% - 15% efficiency CIS cell
- Identify growth parameters for a low resistance tunnel junction.
- Develop a CGS/CIS two-junction tandem solar cell model by AMPS-1D and identify an optimized tandem cell structure with a 25 % efficiency.
- Develop and evaluate a two-step electrodeposition method for growing CGS films.

CuInSe₂ as a Component in High Efficiency Concentrator Photovoltaics

A. Rockett, D.X. Liao, and C.M. Mueller

University of Illinois, Department of Materials Science and Engineering,
1-107 Engineering Sciences Building, MC-233, 1101 W. Springfield Ave., Urbana, IL 61801
arockett@uiuc.edu 217-333-0417



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Epitaxial CuInSe₂ for Ultrahigh Performance Solar Cells

Multijunction solar cells have achieved very high performances with junctions in excess of 1 eV. Ultimate performances can only be achieved with a 1.0 eV cell.

CuInSe₂ is an excellent candidate for this application:

- 1.0 eV energy gap
- Has achieved very high-performance single junctions
- Epitaxial growth has been demonstrated on various surfaces.



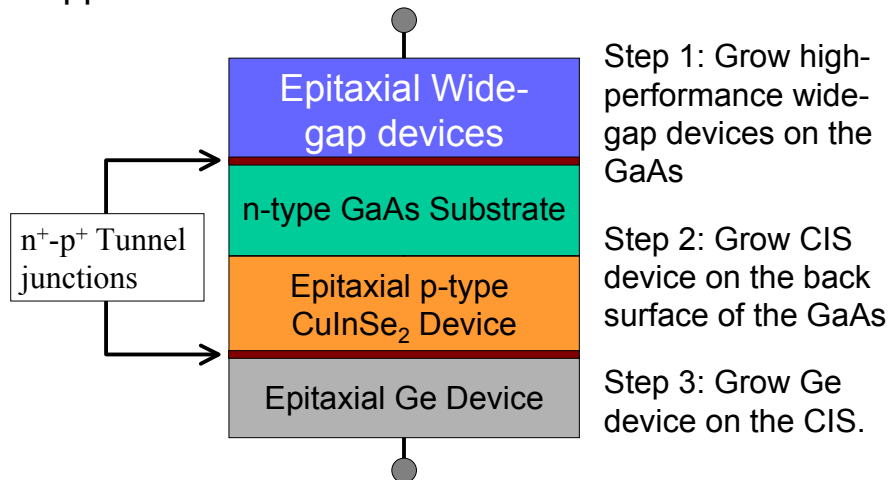
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Plan for High Performance Devices

Approach #1: Growth on a GaAs substrate.



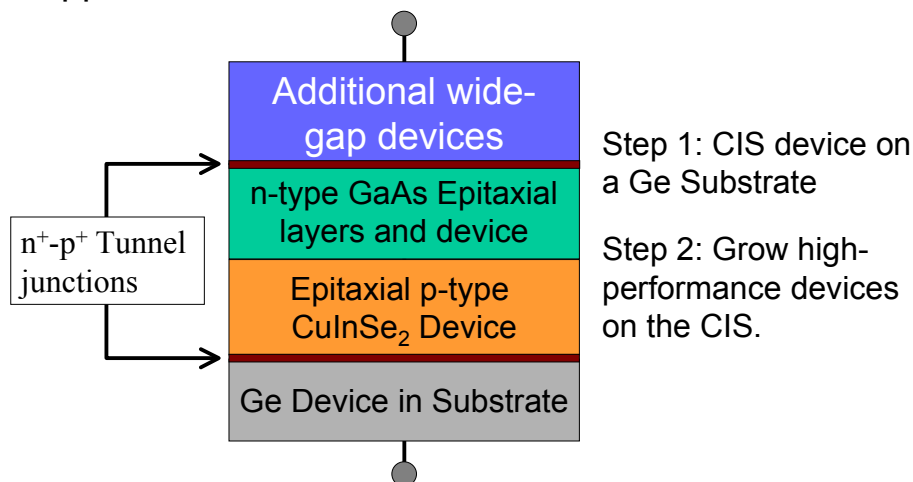
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Plan for High Performance Devices

Approach #2: Growth on a Ge substrate.



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Plan for High Performance Devices

Approach #1: Growth on a GaAs substrate:

Advantages:

- Can begin with a known high performance device
- CIS can be grown on relatively lower quality back surface of the wafer.
- Segregation of impurities from the CuInSe_2 into the Ge can probably be stopped with barriers.
- Ge growth requires a relatively low temperature.

Disadvantages:

- Requires a process that the high-performance devices can withstand.
- Defects are likely in the Ge device.



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Plan for High Performance Devices

Approach #2: Growth on a Ge substrate:

Advantages:

- Indirect Ge requires a thick layer. Substrate is ideal for this.
- All growth on the same side of the substrate.

Disadvantages:

- Potential for detrimental segregation of material from CuInSe_2 into the GaAs can be disastrous.
- Difficult to grow extremely high-performance GaAs on CuInSe_2 . May lead to poor high gap device junctions.



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Experimental Plan

Determine the performance that can be achieved in CIS junctions

on n^+ GaAs:

- Establish the quality of solar cells that can be produced from epitaxial CIS layers using a p-CIS n-GaAs heterojunction for current collection.

on p^+ and n^+ -Ge:

- Establish conditions for epitaxy on Ge.
- Produce solar cells in collaboration with NREL and IEC on these layers.
- Send completed layers to NREL as substrates for high-performance device growth.



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Progress to date:

n^+ GaAs Substrates

- Substrates obtained with various orientations
- Epitaxial layers grown
- Preliminary results show diode-like behavior.

p^+ and n^+ -Ge Substrates:

- Substrates on order.

Additional ultrahigh vacuum deposition system is being put into order for CIS growth to provide cleaner films and larger substrates (for compatibility with NREL growth systems).



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Issues:

n^+ GaAs Substrates

- Growth of Ga-free CIS for 1 eV gap. Ga diffusion from the GaAs substrate could be a problem.
- Deposition conditions exist to minimize diffusion.
- Diffusion within the depletion width of the junction could be a positive effect.
- Need to avoid Cu diffusion into the GaAs.
- Choice of substrate orientation (for junction properties and epitaxial temperature).
- CIS-GaAs heterojunction performance as a solar cell.

p^+ and n^+ -Ge Substrates:

- Past efforts have shown problems with epitaxy on Ge. Extensive work was never performed to determine opportunities.
- And similar issues as above.



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Our “Beyond the Horizons”
program should also contribute to
the Thin Film HiPer program...



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Experimental

Film Growth:

$T_s = 540 - 725\text{ }^{\circ}\text{C}$.

Rate $\sim 1\mu\text{m/hr}$.

Cu-Ga target for Ga-containing alloys

Pure Cu target:

Ga diffuses out of the substrate.

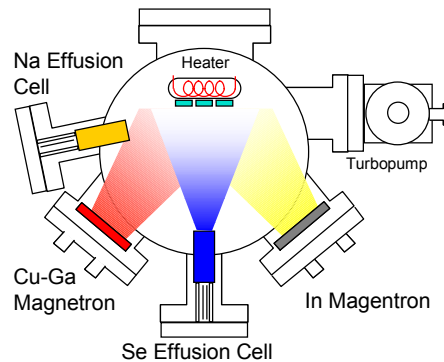
Diffusion rate varies with composition.

Classic Kirkendal voids form at the CIS/GaAs interface.

Cu diffusion into the GaAs prevents conduction in the substrate.

Best epitaxy on (111). Rough surfaces on (110).

$T_s = 540\text{ }^{\circ}\text{C}$ (220); $640\text{ }^{\circ}\text{C}$ (100); and $700\text{ }^{\circ}\text{C}$ (112)

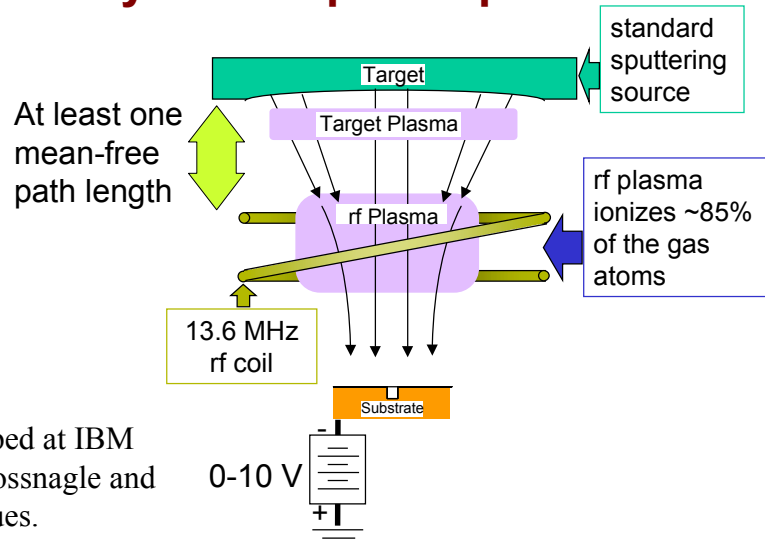


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Ionized Physical Vapor Deposition



Developed at IBM
by S. Rossnagle and
colleagues.

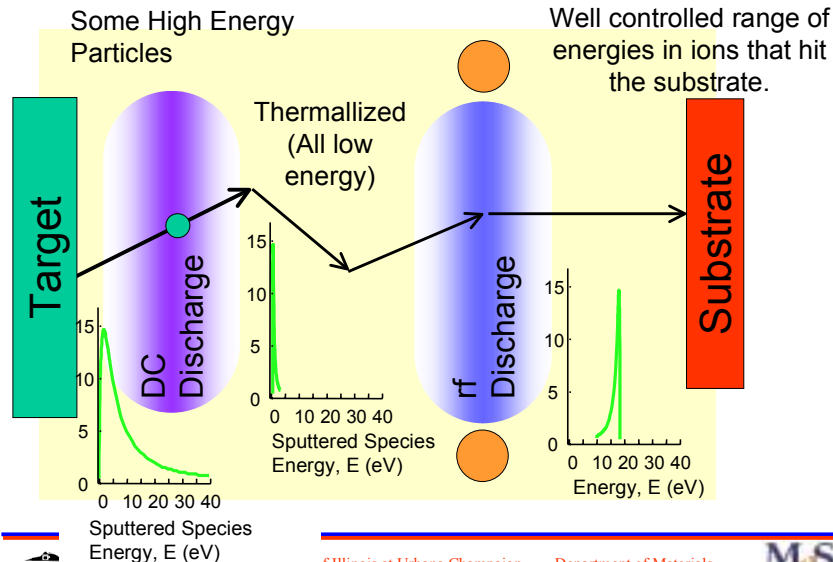


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Ionized Physical Vapor Deposition



Sputtered Species
Energy, E (eV)



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Ionized Physical Vapor Deposition

Advantages:

- Simple add-on to existing deposition environments
- Most atoms hit the surface with moderate kinetic energy
- Energy can be controlled to be below the damage threshold but well above the energy needed for surface diffusion and cluster disruption.
- By control of the atom impact energy we can make atoms go where they normally would rather not go without forcing them into defect sites.
- Supply of energy to the growth surface can replace the energy normally supplied by heating.



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Experimental Plan

Modify existing deposition systems for ionized physical vapor deposition:

- Existing growth conditions can be used and normal growth continued to show deposition condition stability.
- Add rf coil in front of the substrate in both the existing and the new ultrahigh vacuum deposition systems.
- After establishment of growth conditions, turn on the rf power and show the changes which can be achieved in deposition conditions.
- Produce solar cells from resulting materials for comparison with existing results.



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Current Status

- Contract in place.
- New graduate student hired
- Training in progress
- Parts being assembled for new ultrahigh vacuum deposition system to increase materials production capability



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High-Performance PV Kickoff Meeting (NREL, Denver, 18 Oct 2001)

**InGaP/GaAs-on-Ceramic
Thin-Film Monolithically Interconnected,
Large-Area, Tandem Solar Cell Array**

Phase I:

**"Development of Low-Cost Substrates and Deposition Processes
For High-Performance GaAs-Based Thin-Film Solar Cells"**

MICHAEL G. MAUK, BRYAN W. FEYOCK, and JEREMY BALLIET

AstroPower, Inc.

Solar Park, Newark, Delaware USA 19716-2000

Tel: 302-366-0400 ext 133

Fax: 302-368-6474

e-mail: mauk@astropower.com

website: www.astropower.com

Broad Objective

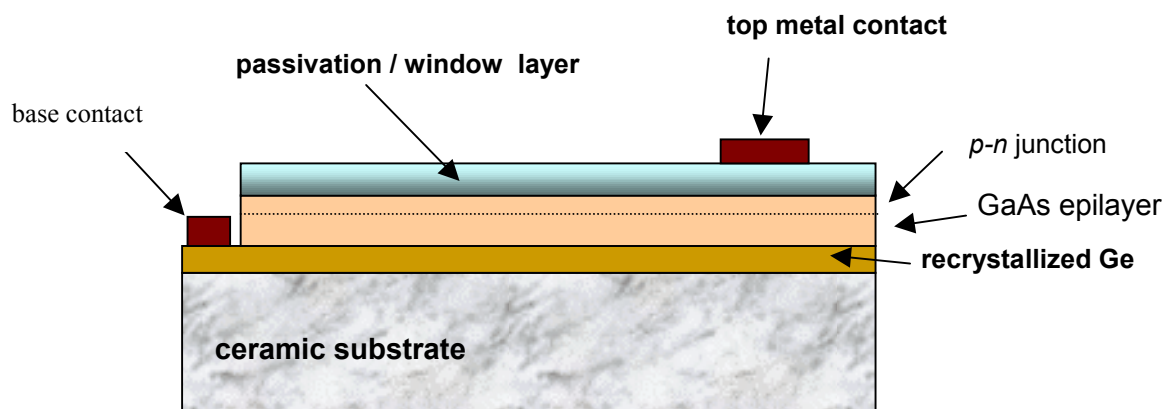
To transfer space solar cell technology that currently yields record-efficiency solar cell performance to a low-cost structure for terrestrial applications

Basic Approach

- replace the expensive single-crystal Ge (or GaAs) substrate wafer with a cheaper alternative
- develop low-cost, high-throughput epitaxy / deposition processes in place of MOCVD
- achieve high efficiencies in GaAs/InGaP solar cells based on thin-films of large-grain ($> 1\text{mm}$) polycrystalline GaAs (and Ge)

Short-Term Goals

- recrystallize Ge films on alumina ceramic substrates to yield highly-oriented, smooth, low-defect layers with grain sizes $\gg 1\text{ mm}$
- characterize Ge films electrically and structurally
- use Ge-on-ceramic structures as substrates for GaAs epitaxy. Ge functions as a "seeding layer" for epitaxial growth of GaAs



Criteria for High-Efficiency GaAs-Based Solar Cells

- Close thermal expansion match between substrate / GaAs
- Grain Size > 1 mm
- Close (< 1%) Lattice Matching of Epilayers
- Suppress Silicon and Germanium Autodoping (from Substrate)
- Defect Densities (< 10^5 cm^{-2})
- Suppress antiphase domains (GaAs/Si or GaAs/Ge)
- Front surface passivation or wide-bandgap window layer
- Back surface field or wide-bandgap cladding layers
- Possibly include light trapping (back reflector or front texturing)

New Features

- 1-micron thick undoped "space-charge" layer to reduce tunneling currents [Research Triangle Inst.]
- post-growth grain boundary passivation (plasma H^+ treatments, sulfur or selenium treatments [Milnes et al.], Zn diffusion in grain boundaries)
- all-top contact design for monolithic interconnection (exploiting electrically insulating ceramic substrate)
- buried conducting layer (heavily-doped Ge)

Emitters/Front Surface Passivation

- AlGaAs wide-bandgap "window" layer (LPE or MOCVD) [HOVEL]
- InGaP emitter or window (LPE, MOCVD, CVD, CSVT)
- Thin (<100 Å) homojunction [FAN]
- ZnSe emitter or window [BLISKE]
- GaAsP emitter or window [BARNETT AND PAREKH]
- Phosphorus "exchange" reaction $\text{P} + \text{GaAs} \rightarrow \text{GaAsP}$
- Aluminum "exchange" reaction $\text{Al} + \text{GaAs} \rightarrow \text{AlGaAs}$
- Anodic oxidation, SiNx CVD
- Chemical surface treatments (Se, S, etc..)
- Zn-diffused, anodic oxide emitters [SULIMA]
- Transparent Conducting oxides for current spreadings and ARCs [COUTTS]
- Plasma-enhanced CVD of SiC [IEC]

Ceramic Substrates

- Mixtures of $\text{Al}_2\text{O}_3/\text{SiO}_2$ for TEC match to GaAs/Ge
- Impurity effects
- Smoothness, wetting properties of molten Ge

Epitaxy Options

GaAs: CSVT or CVD with H_2O , I_2 or HCl as transport agent

Ge: CSVT or CVD with H_2O , I_2 or HCl as transport agent

ZnSe (as a passivating layer): CVT with H_2

GaAs, AlGaAs, InGaP, GaAsP

- LPE
- Hydride or Halide CVD
- MOCVD by outside group

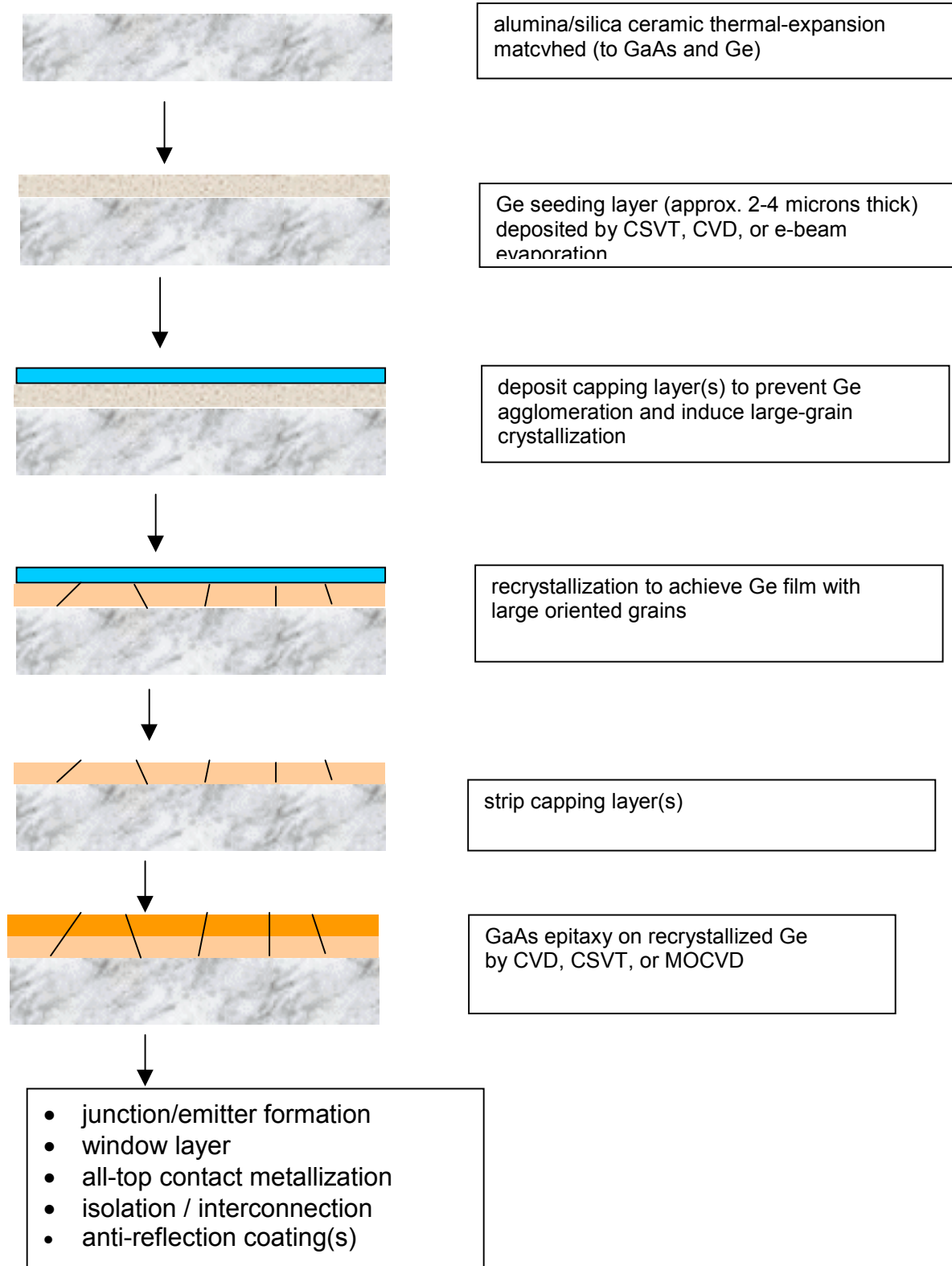
Some Criteria for Epitaxy

- Large scale: substrate size $> 10\text{-cm} \times 10\text{-cm}$
- Atmospheric pressure operation
- No toxic precursors or by-products
- Inexpensive precursors
- Potential for continuous-mode operation
- Fast deposition rates (> 1 micron/min)
- Low capital equipment cost
- Reasonable control and instrumentation
- In-situ diagnostics
- Avoid halogens (corrosion problems, unwanted co-transport)

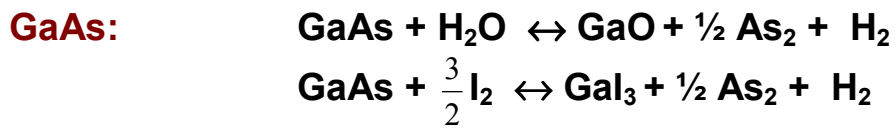
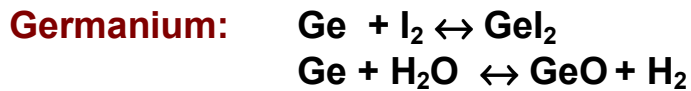
Some Related Applications

- InGaP / Si cells for photoelectrochemical hydrogen generation
- Large-area LEDs and detector arrays
- Space solar cell arrays

Thin-Film Poly GaAs-on-Ceramic Solar Cell Process



Reversible-Transport Reactions for Deposition of Silicon, Germanium, and GaAs and Their Applications to Solar Cells



Advantages of Reversible Transport Reactions for Si, Ge and GaAs CVD

- Fast Deposition Rates (5 to 10 microns/min)
- Transport agent stored as a solid or liquid
- Recycled Transport Agent
- Atmospheric-Pressure Operation
- No Highly Toxic Precursors or Byproducts
- High-Deposition Efficiency (esp. Close-Spaced Configuration)
- Simple Control (source temp, substrate temp, transport agent conc.)
- Doping from Source

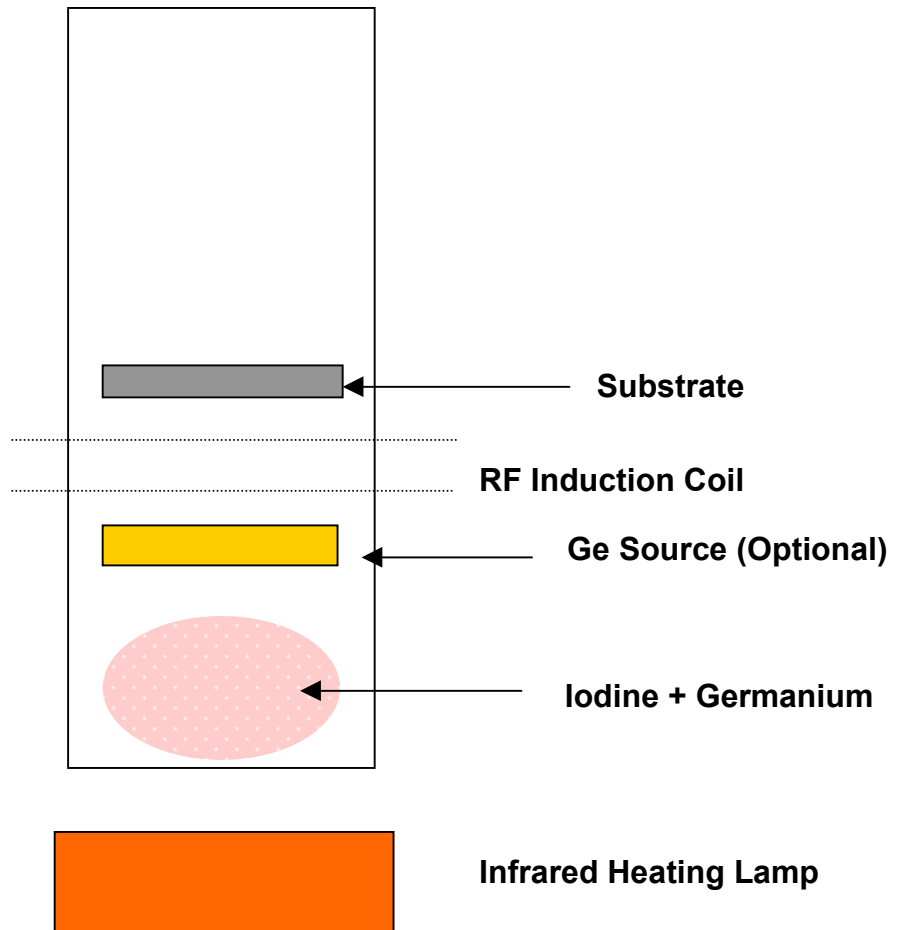
Silicon and Germanium

- Annealed / Recrystallized
- Post-Growth Gettering
- Less-Pure Solid Sources

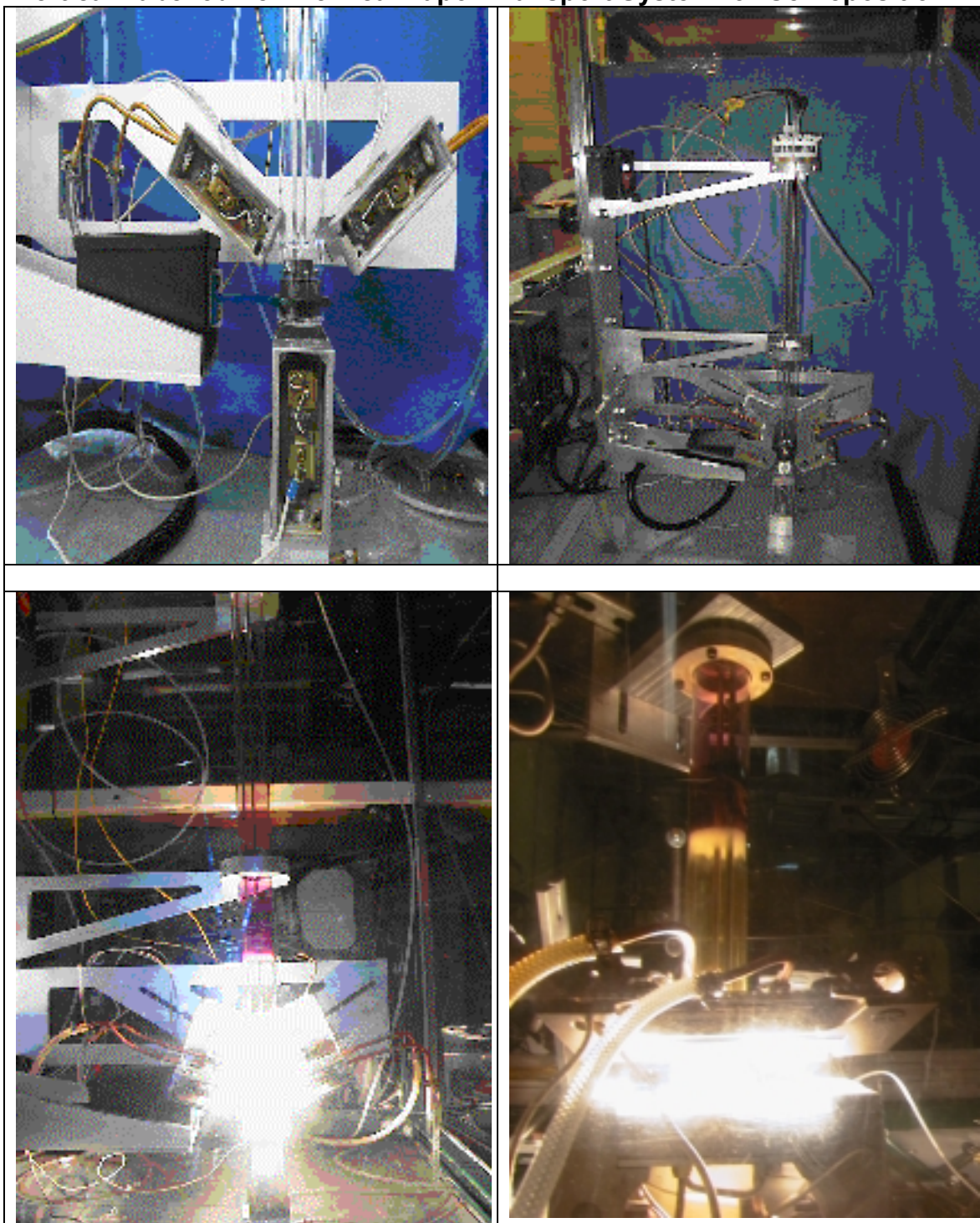
Issues

- Semiconductor Purity
- Co-Doping (from substrate)
- Grain-Size and Texture, Electrical Activity of Grain Boundaries

Vertical Tube Iodine Chemical Vapor Transport System for Depositing Ge on Ceramic Substrates

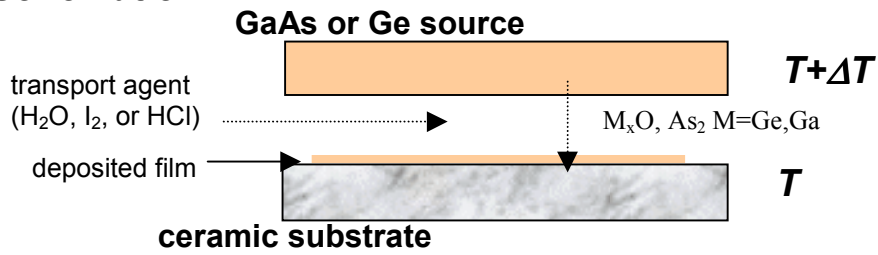


Vertical Tube Iodine Chemical Vapor Transport System for Ge Deposition

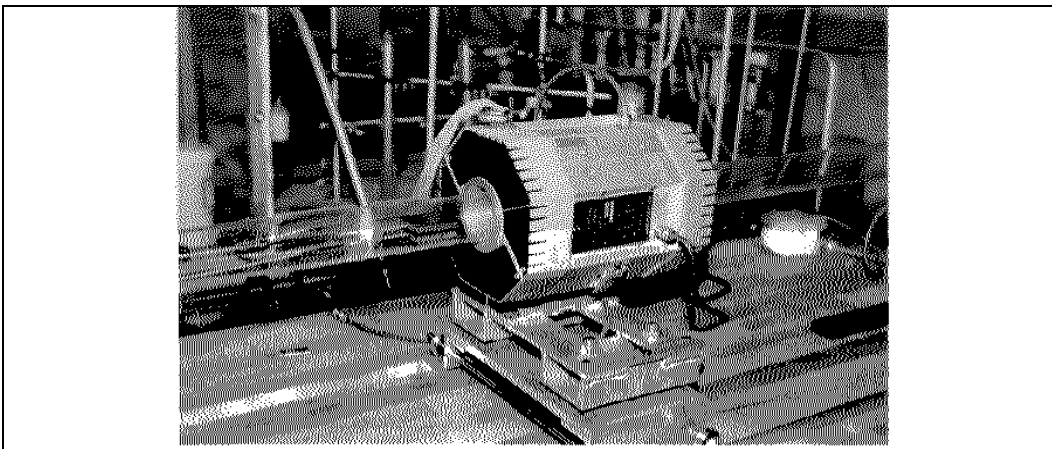
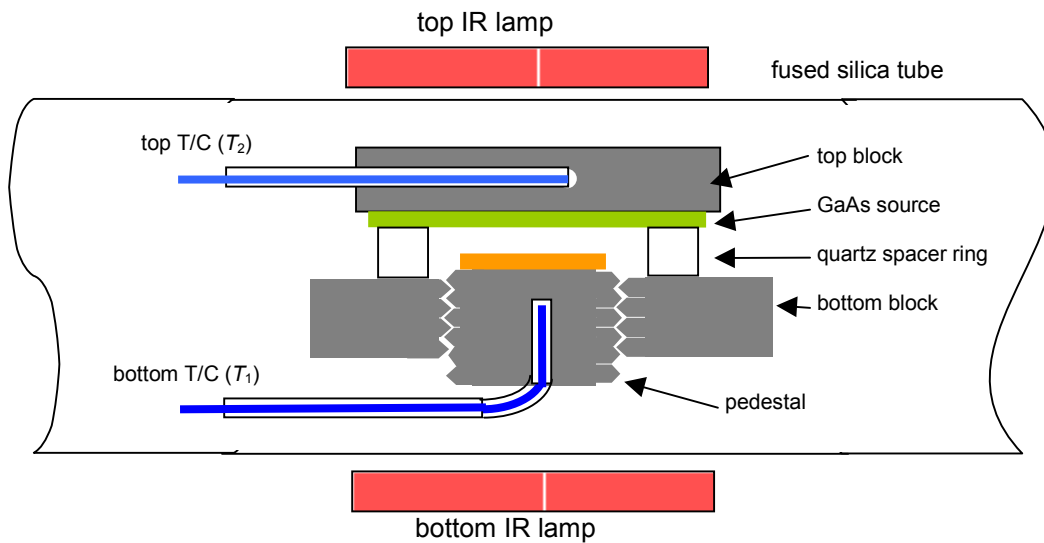


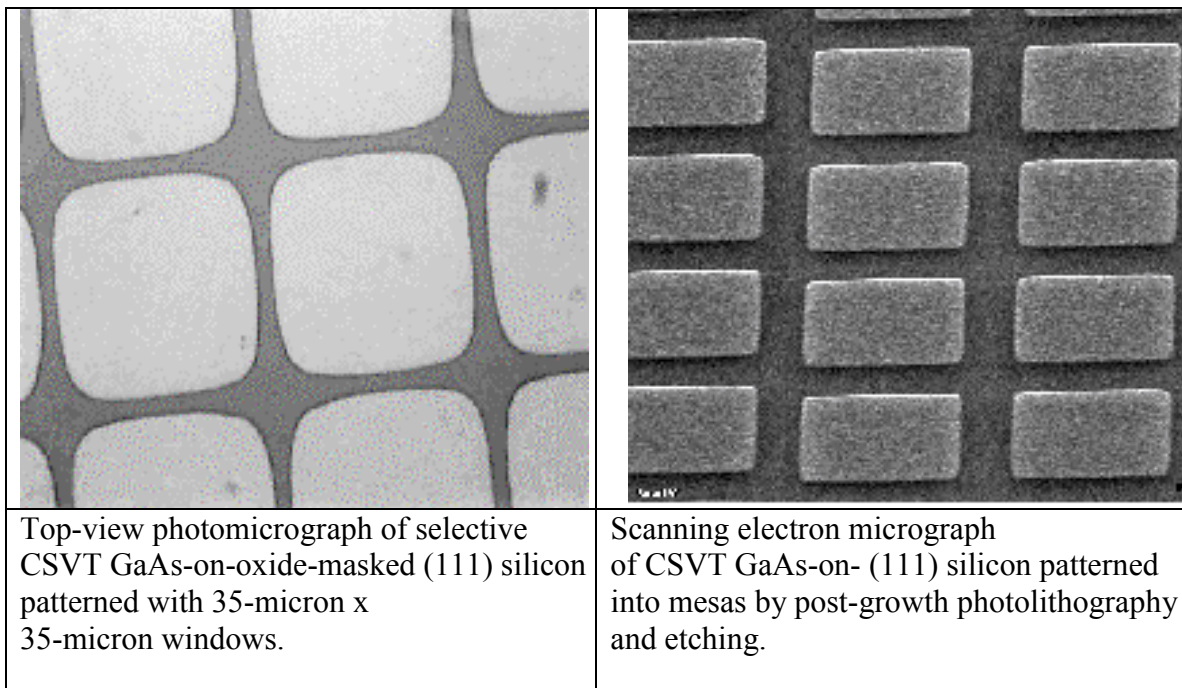
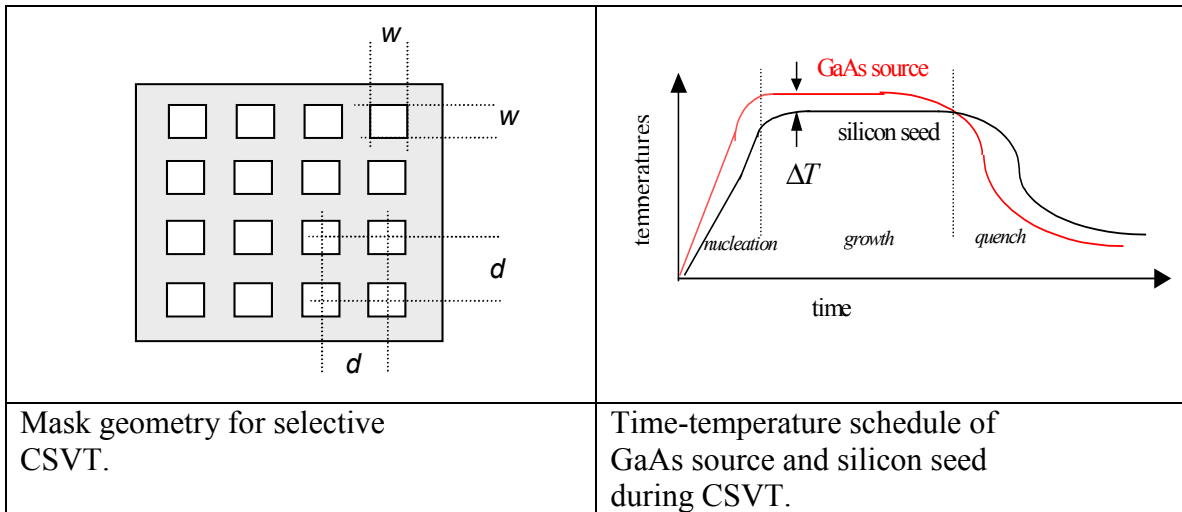
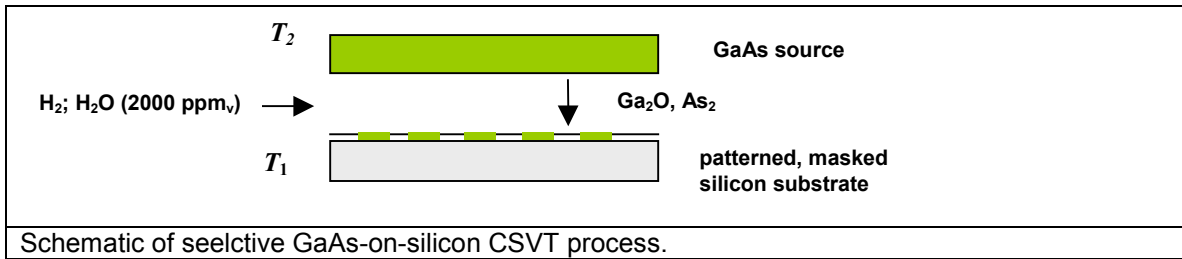
CSV (Close-Spaced Vapor Transport)

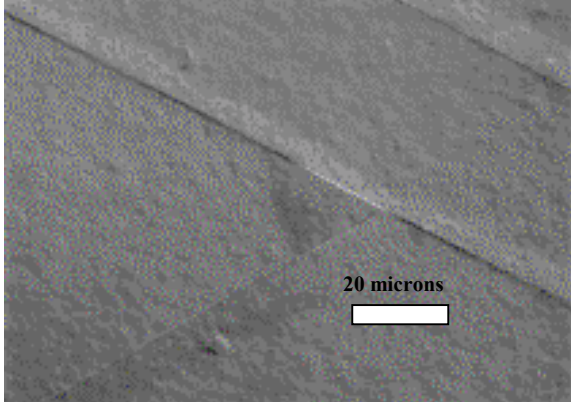
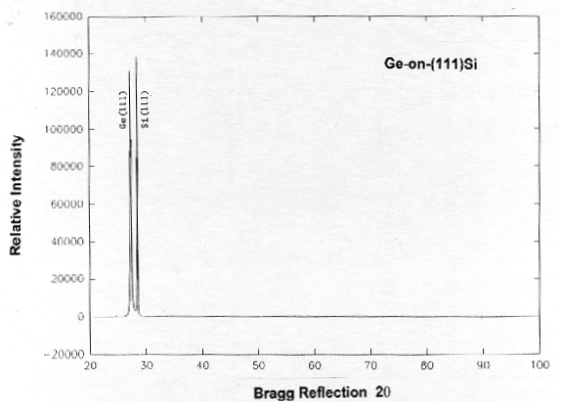
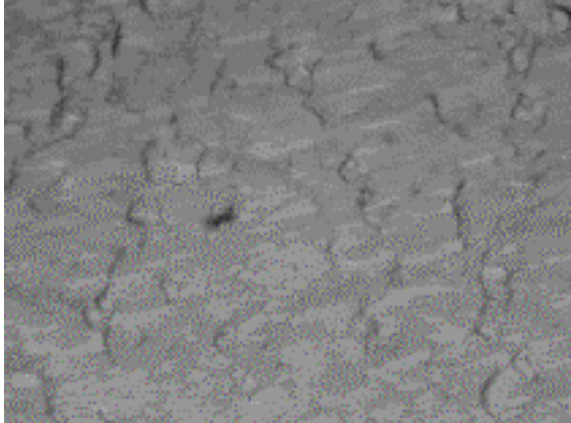
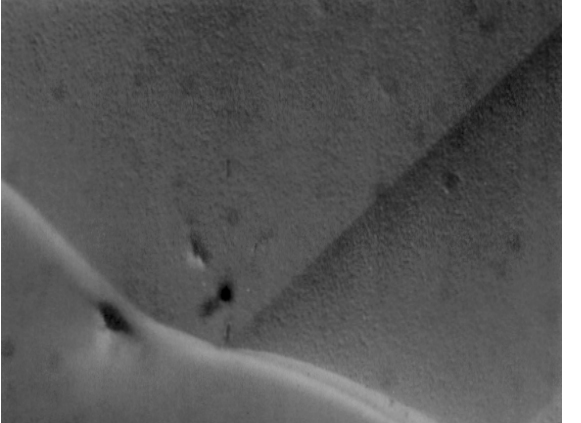
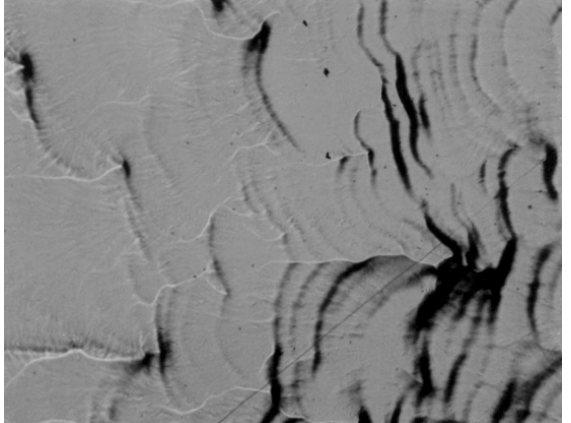
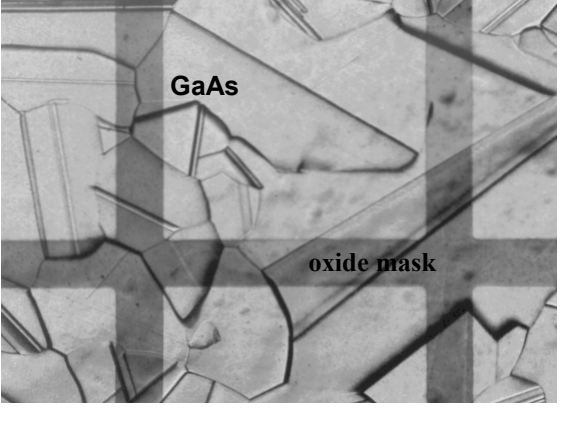
Schematic

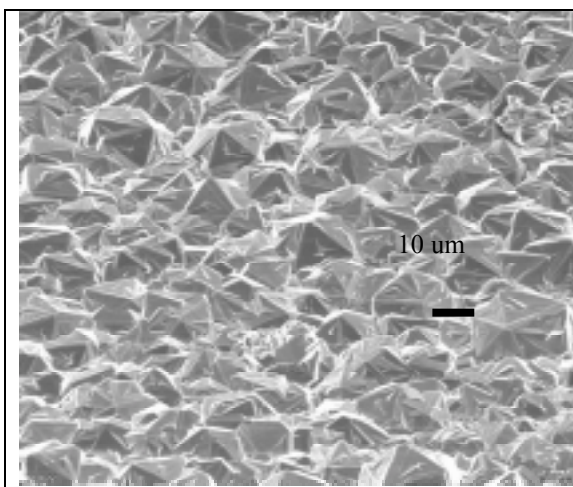


Detail

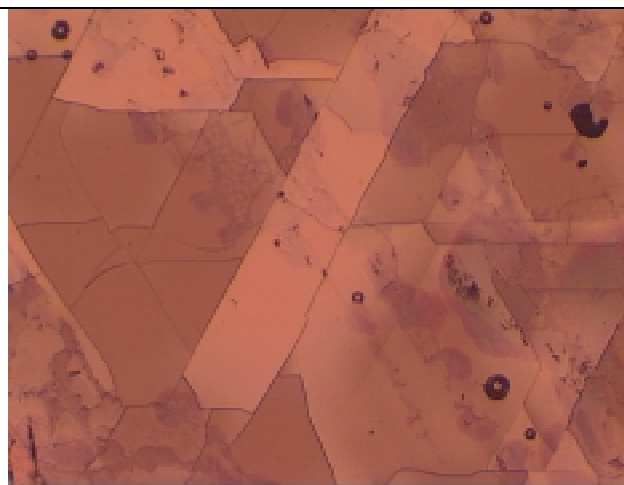




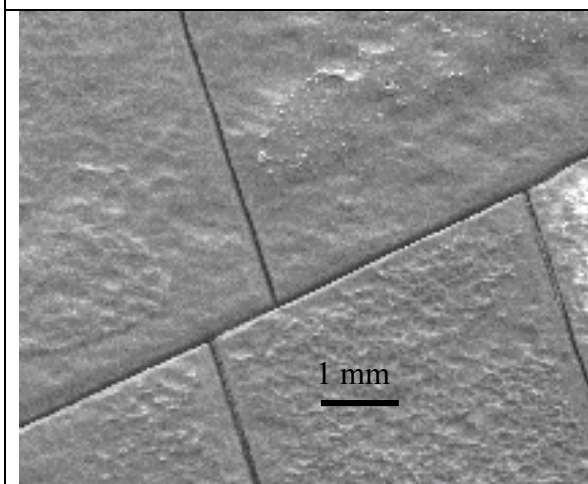
	
<p>Single-crystal Ge epitaxial film grown on (111) Si by a Close-Spaced Vapor Transport (CSVT) process at 700 °C using water vapor as the transport agent.</p>	<p>X-ray diffraction spectra of Ge-on-Silicon sample.</p>
	
<p>GaAs epitaxial film on polycrystalline Ge substrate grown by CSVT at 700 °C using water vapor as the transport agent.</p>	<p>Top-view photomicrograph of a 2.5-micron thick Ge layer grown by CSVT on a Silicon-Film™ polycrystalline silicon sheet substrate.</p>
	
<p>Top-view photomicrograph of 4-micron thick Ge layer grown on a Silicon-Film™. Ge layer has been melted and recrystallized by annealing the sample at 950 °C in hydrogen for one-hour.</p>	<p>Top-view photomicrograph of selective CSVT GaAs on a Silicon-Film™ polycrystalline silicon sheet substrate. Squares are 500 microns x 500 microns.</p>



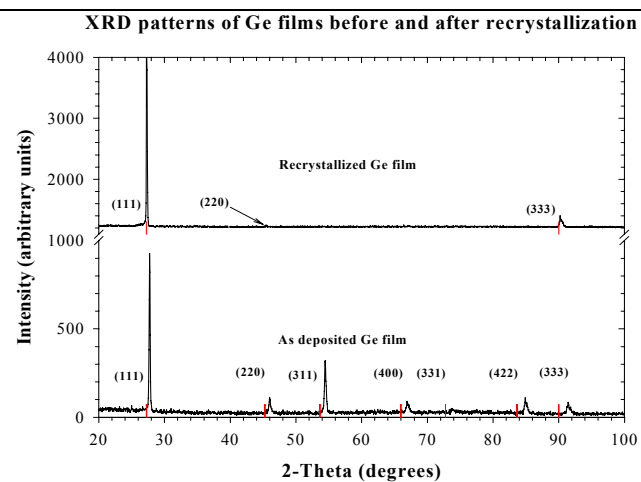
Scanning electron micrograph of as-deposited Ge film on alumina ceramic substrate.



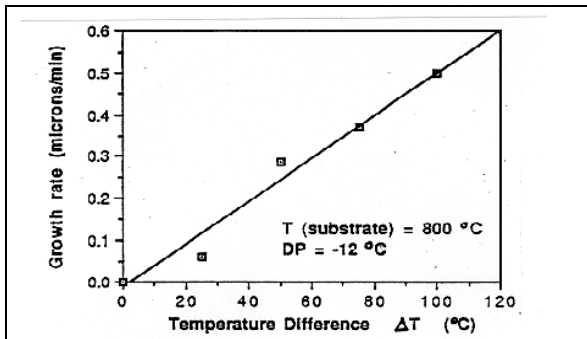
Top-view photomicrograph of recrystallized 5-micron thick Ge film on fused silica substrate.



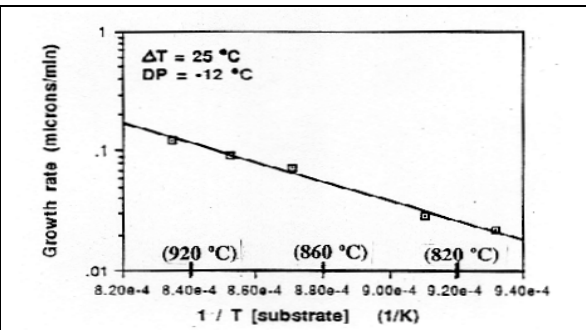
Scanning electron micrograph of recrystallized Ge film on fused silica substrate.



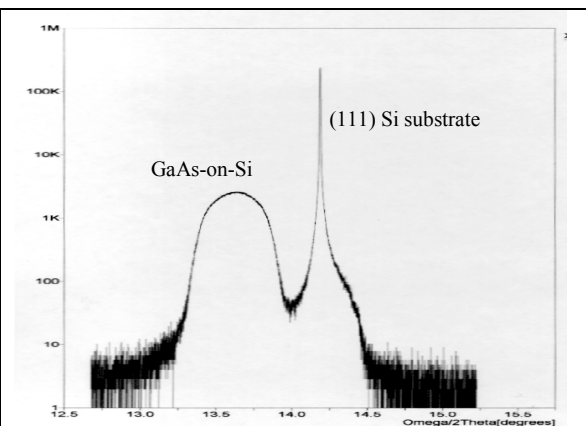
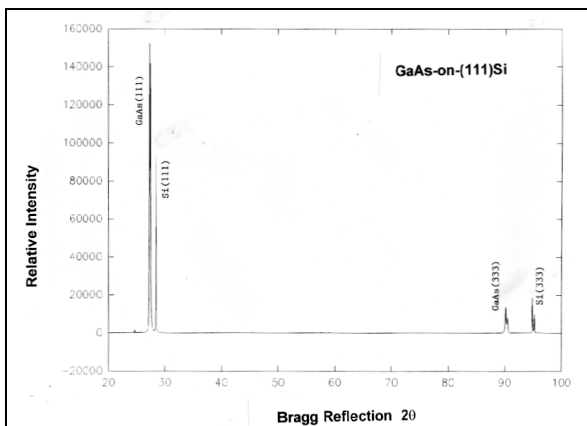
X-ray diffraction of Ge films before and after recrystallization.



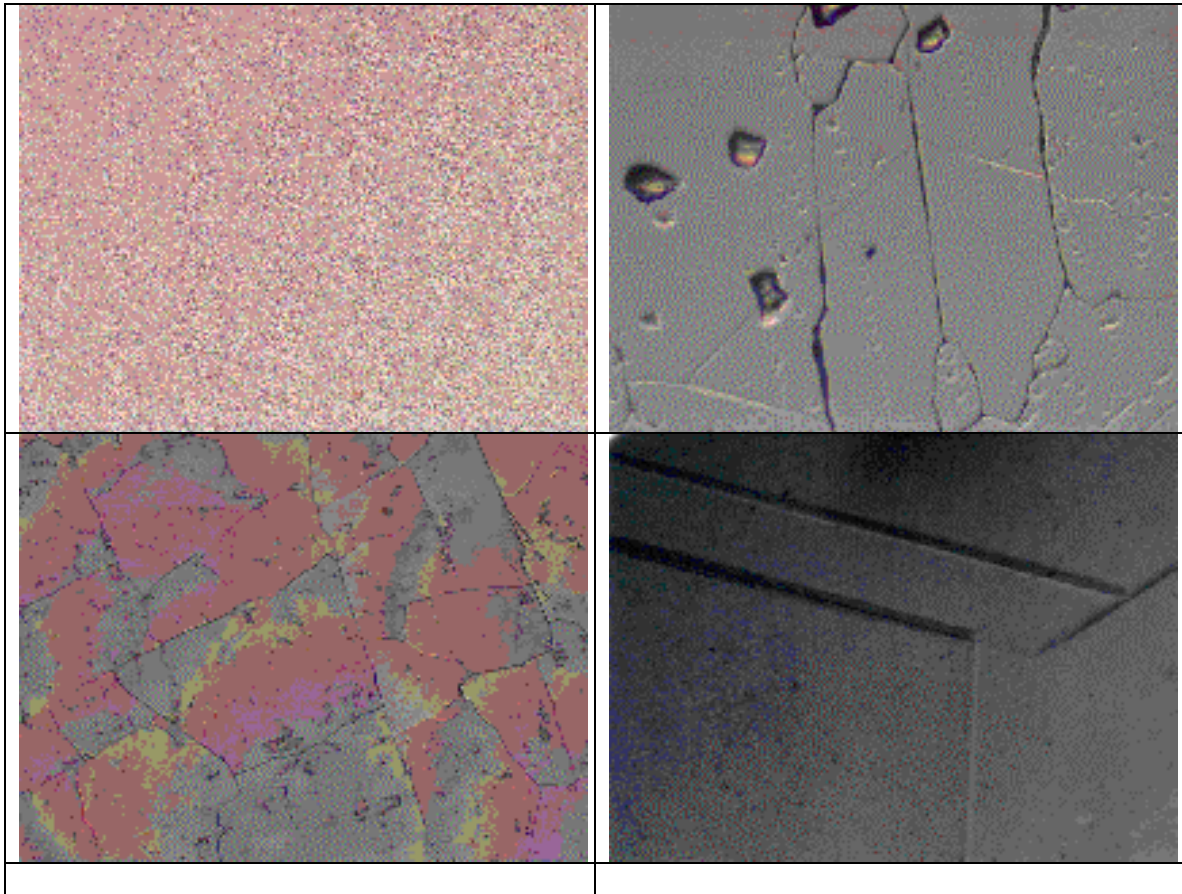
GaAs-on-Si CSVT kinetics as a function of temperature difference ΔT between source and seed.



Log of growth rate vs. reciprocal of substrate temperature for GAAS-ON-Si CSVT.



X-ray diffraction spectra for 0.4-micron thick GaAs-on-silicon films made by CSVT. **a.** wide scan, low resolution, **b.** double-crystal rocking curve.





NREL Basic Research Toward a 40% Efficiency

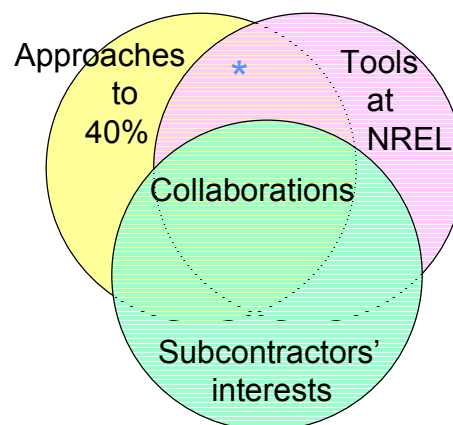
Sarah Kurtz

High-Performance PV Kickoff Meeting

October 18, 2001

Operated for the U.S. Department of Energy by Midwest Research Institute • Battelle • Bechtel 

Where we fit in the big picture

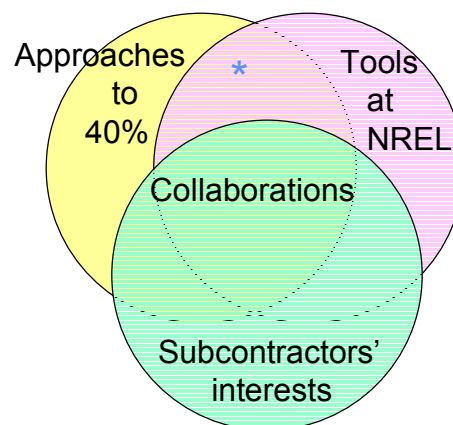


Opportunities for collaborations between NREL and subcontractors

- TEM - characterization of mismatched layers
 - Mowafak_AlJassim@nrel.gov (303)384-6602
- Cell and module measurements
 - Keith_Emerly@nrel.gov (303)384-6632
- Characterization (TRPL, DLTS, FTIR, SIMS, etc.)
 - www.nrel.gov/measurements
 - Pete_Sheldon@nrel.gov (303)384-6533
- Resource assessment (spectrum of direct beam) and how it affects cell design (AM1.5Direct is *not* appropriate for outdoor concentrator measurements)



Where we fit in the big picture



Dilute-nitrogen alloys

(GaInAsN, etc.)

(one of many possible approaches)

Problem = Photovoltaic quality is degraded by nitrogen

Reasons for studying this approach:

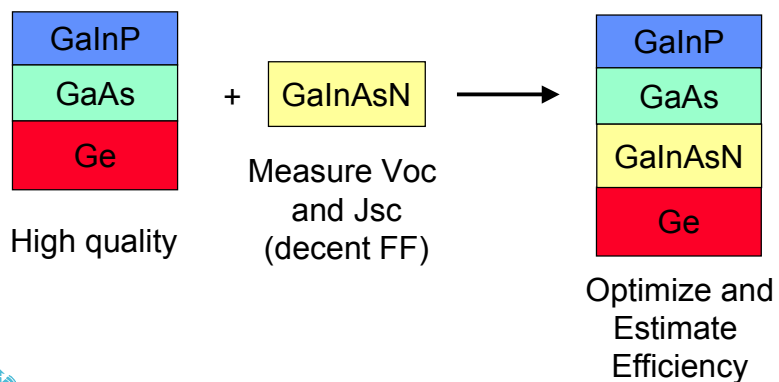
- GaInP/GaAs/GaInAsN/Ge theoretically > 50%
- GaInP/GaAs/GaInAsN/Ge is similar to GaInP/GaAs/Ge
- “best” GaInAsN cells are close to “break even”
- This basic research problem is well matched to



NREL's mission and strengths

Breakeven calculation - approach

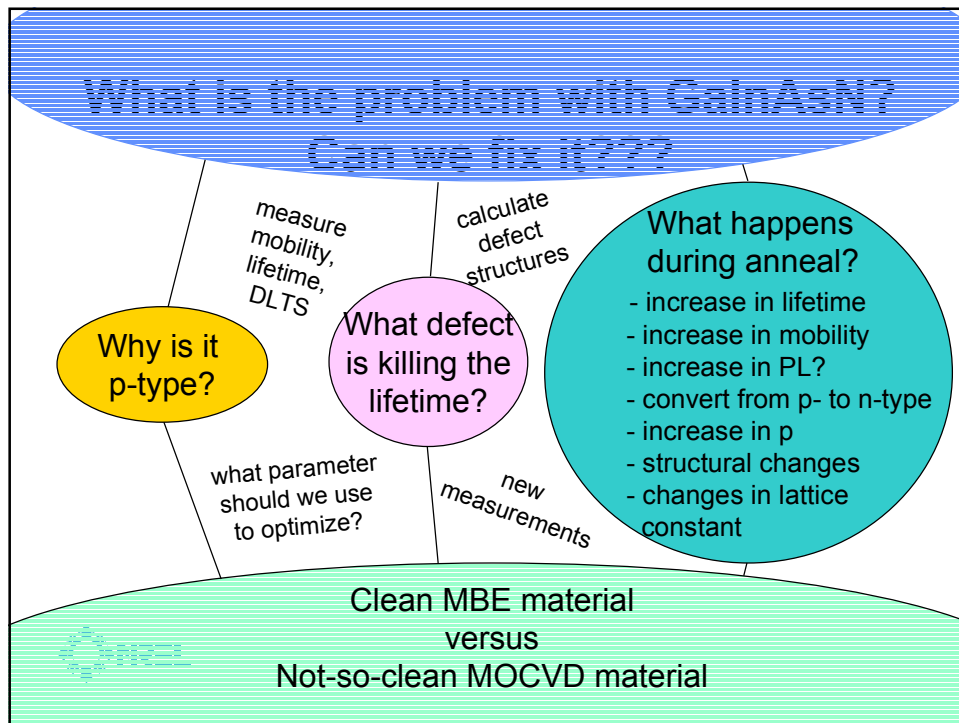
(Friedman)

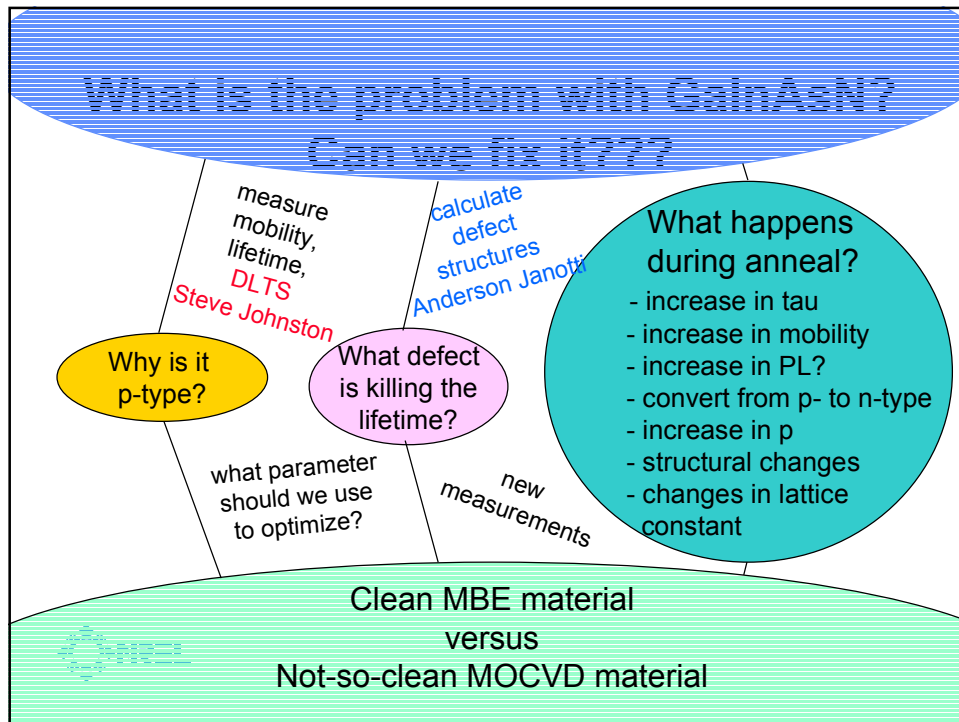


Breakeven calculation - results (Friedman)

Conditions	Expected 3-junc. Efficiency (no series es.)	Estimated 4-junc. Efficiency with best GaInAsN
AM0 - 1 sun	29.9%	28.2%-29.4%
AM1.5 Global (500 suns)	40.2%	38.6%-39.7%
AM1.5 Direct (500 suns)	38.2%	39.1%-40.5%

Conclusion: **optimist** - we're almost there
pessimist - it's not hopeless





Discussion topics

- Collaboration on materials measurements
- Spectral issues
- Stability/degradation issues (heat, high current, intense light; will it last for 30 years?)





National Center for Photovoltaics



Reference Conditions for PV Concentrators?

Keith Emery

National Renewable Energy Laboratory

1617 Cole Blvd., Golden CO 80401

email: keith_emery@nrel.gov

MEASUREMENTS & CHARACTERIZATION DIVISION
www.nrel.gov/mcd



National Center for Photovoltaics

Concentrator Standards

Cells - From ISE, NREL, Sandia, Progress in PV

25 °C, 1-sun = 1000 W/m² total irradiance

ASTM E891 direct normal spectrum (767 W/m² direct)

Area = total area minus peripheral bus bars

Modules -

Area = lens/mirror aperture area

Prevailing spectral irradiance, 850 W/m² total direct irradiance, 20 ° air temperature

(ASTM draft E130, PVUSA, IEEE 1513)

Prevailing spectra corrected to AM1.5, 850 W/m²

direct irradiance, 150 W/m² diffuse irradiance, 50 °C

cell temperature, (IEEE 1513)

MEASUREMENTS & CHARACTERIZATION DIVISION
www.nrel.gov/mcd





Outline - What is the Problem?

The direct normal reference spectrum is not representative of locations where concentrators might be deployed?

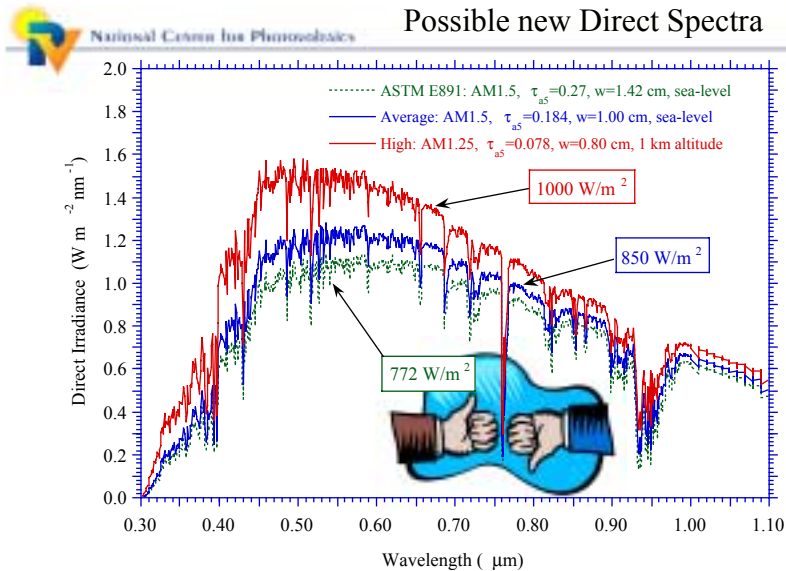


When 2-axis global 1001 W/m² ($\sigma=1.3$)
834 W/m² direct irradiance ($\sigma=22.8$)
24.4 °C air temperature ($\sigma=4.0$)
4.4 m/s wind speed ($\sigma=1.1$)
0.08 turbidity @ 500 nm ($\sigma=0.02$)
1.43 relative optical Air Mass ($\sigma=0.09$)
37 sites in southwest U.S. 433,562 observations
D. Myers et al., *Proc. 28th IEEE PVSC.*, p. 1202, 2000.



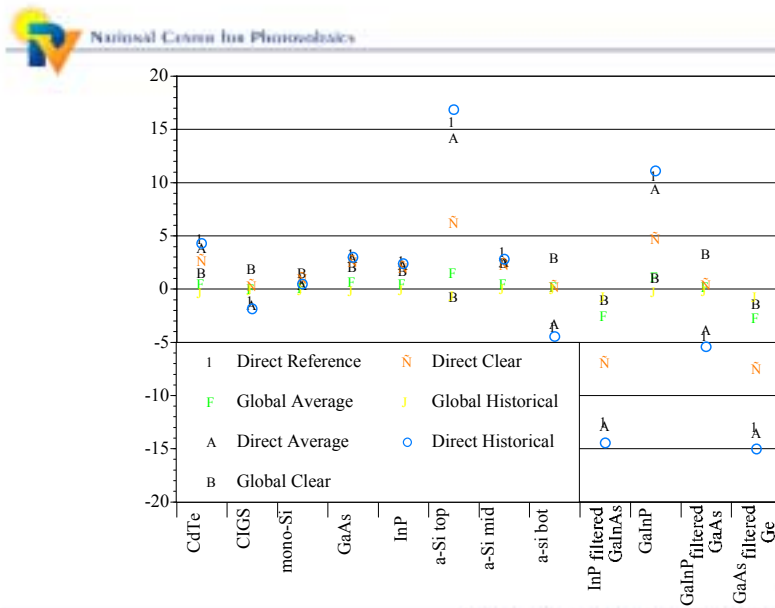
Not enough “blue” in ASTM E891 direct because turbidity of 0.27 used.
Total irradiance of 767W/m² for E891 is too low



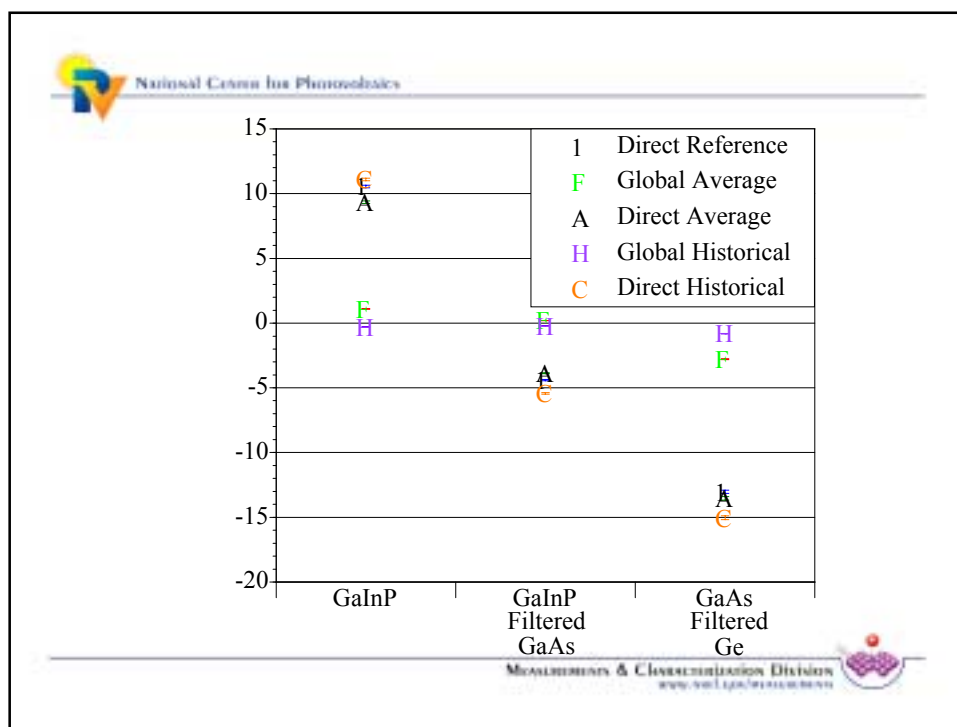
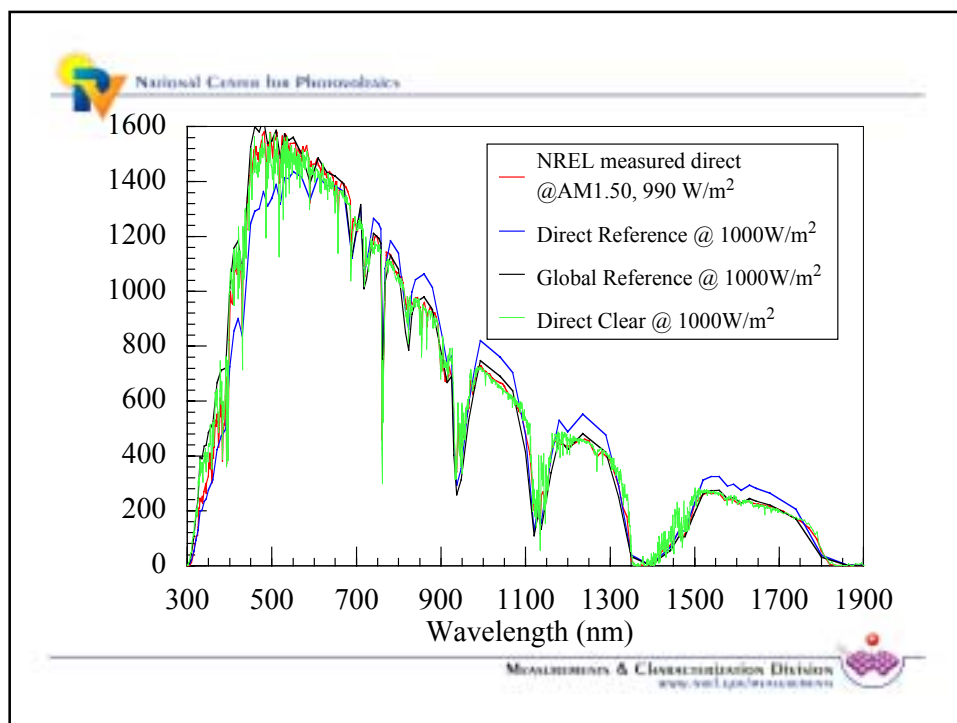


D. Myers, C. Gueymard, others this conference, draft ASTM, Journal

Measurements & Characterization Division
www.nist.gov/mcdiv



Measurements & Characterization Division
www.nist.gov/mcdiv





The spectral sensitivity of single junction devices (Si, GaAs) is not more than $\pm 2\%$ under a wide variety of conditions.



In “real world” concentrator modules GaInP/GaAs/Ge cells optimized for the AM0 or global spectrum consistently outperform cells optimized for the direct spectrum



Concentrator cell
34.0% for global
30.6% for direct
spectrum





Where do we go from here?

Concentrator modules will be evaluated outdoors under prevailing clear-sky conditions without a correction to the existing direct reference spectrum. At most a correction to AM1.5 (IEEE).

Cells for concentrator modules should be optimized for maximum module efficiency (direct average or global reference).

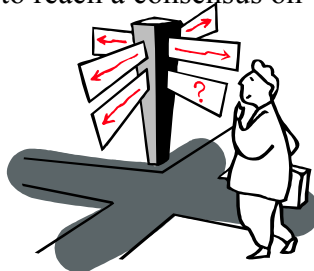
Report concentrator cell efficiencies under the global and direct reference spectrum.



Where do we go from here?

Get the standards labs together (in person or via email) to make a decision to change the direct reference spectrum to reflect a more realistic turbidity.

Get the U.S. concentrator industry and the national labs (Sandia and NREL) to reach a consensus on what to do.



High Performance, Low Cost III-V PV Concentrator Module

Raed A. Sherif & Richard R. King

October 18, 2001

Goal & Work Scope

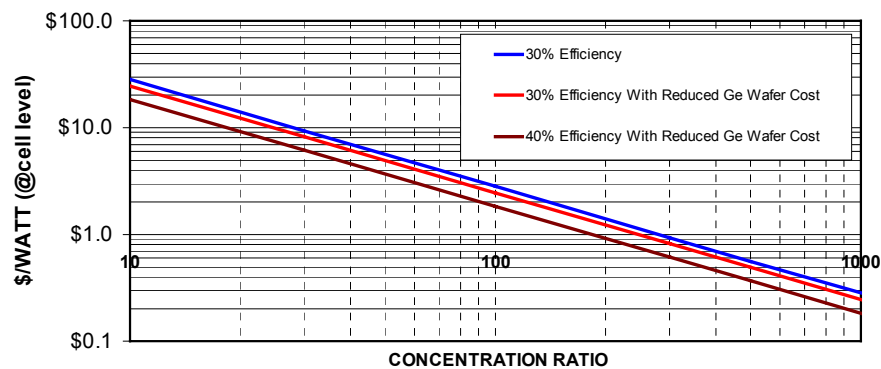
- Identify the critical paths towards achieving a 33% module conversion efficiency at \$1 per Watt.
- A two-year effort investigating:
 - How to increase III-V PV cell conversion efficiency
 - How to make III-V PV cells more robust for high concentration
 - How to reduce the manufacturing costs of III-V PV cells
 - Module designs that are low cost, yet reliable, and have low thermal resistance to the cooling fluid.

Cost Reduction Approaches for III-V PV Cells

SPECTROLAB

A BOEING COMPANY

1. Increase PV cell conversion efficiency & reduce its manufacturing cost
2. Use higher concentrations



Specific Objectives in this framework

SPECTROLAB

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- PV Cell
 1. Develop a high efficiency (> 32%) monolithic multijunction PV concentrator cell
 2. Develop a very high efficiency (> 35%) integrated multijunction PV concentrator cell using two or more different substrates.
 3. Develop a robust cell structure and metallization scheme that eliminates cell shunting under high concentration.
 4. Research low-cost substrates for high efficiency multijunction cell growth.
 5. Added task: deliver 100 concentrator cells (~ 1x4 cm) to operate at ~ 440 suns average with maximum concentration of ~ 1600 suns on 1/4 of the cell.
- PV Receiver
 1. Identify designs that have potential for low cost, high performance and good reliability
 2. Build prototypes and demonstrate their performance and reliability

High Efficiency Cell Development Work Scope

SPECTROLAB

A BOEING COMPANY

- Develop a high efficiency concentrator cell (> 32%) by leveraging PV technology used in space applications.
 - Triple junction cells: GaInP/ GaAs/ Ge
 - Metamorphic triple junction cells: GaInP/ GaInAs/ Ge
 - Four junction cells: GaInP/ GaAs/ GaInNAs / Ge
- Demonstrate new prototype cell designs using an integrated cell approach (wafer bonding or mechanical stacking) to achieve ultra-high cell efficiency (> 35%)
 - Baseline: GaInP/ GaAs/ Si
 - Additional examples: GaInP/ GaAs/ Si/ GaSb
 - GaInP/ GaAs/ GaInPAs/ GaInAs
 - Two terminal, 3-terminal, and 4-terminal MJ cell designs

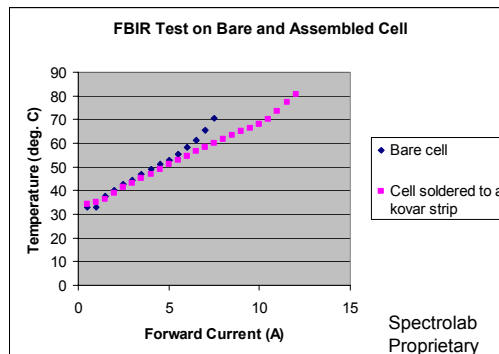
Spectrolab Proprietary

Increased Cell Robustness- Interaction with Thermal Management

SPECTROLAB

A BOEING COMPANY

- Investigate metallization structures that can increase cell robustness to shunting.
- Investigate impact of highly non-uniform flux on a cell level
- Investigate interaction between how a cell is cooled and its current carrying capability.



Module Design

SPECTROLAB

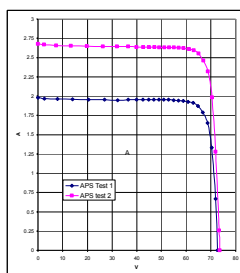
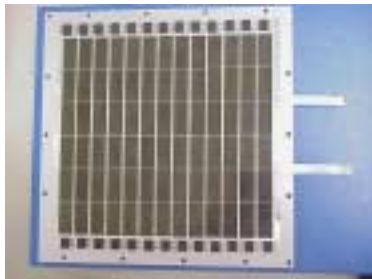
A BOEING COMPANY

- Goal is to have a module design that is:
 - Low cost
 - Enables high concentration
 - Enables mechanized assembly, inspection, and test.
- Identify tradeoffs in module cost, thermal management, and thermal expansion mismatches.
- Investigate different types of designs:
 - Single cell vs. densely packed array designs
 - Active vs. passive cooling
 - Which design(s) are more compatible with mechanized assembly, inspection, and test.
- Define an appropriate test vehicle for testing cells under high concentration.

A Dense PV Array for 2 kW Output at 300 sun Concentration

SPECTROLAB

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Collaboration between Spectrolab, APS,
and Concentrating Technologies

Tasks & Schedule for Cell Development (Year I)

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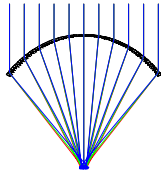
- | | |
|--|---------|
| • MJ cell electrical modeling input for cell growths | Nov '01 |
| • ~150 cm ² of conc. cells delivered to ENTECH
(100 cells with ~1.5 cm ² aperture area) | Dec '01 |
| • Receive alternative low-cost Ge substrates | Jan '02 |
| • GaInP / GaAs / Ge and
GaInP / GaInAs / Ge cell sample delivery | Mar '02 |
| • GaInP / GaAs cell bonded to Si substrate
sample delivery | Jul '02 |
| • GaInP / GaAs cells without metallization for
collaboration with NREL on mech. structures | Jul '02 |

Tasks & Schedule for Receiver Development (Years I&II)

SPECTROLAB

A BOEING COMPANY

- | | |
|--|---------|
| • Thermal, thermal stress, electrical and cost modeling
for concentrator cells and modules. | Nov' 01 |
| • Preliminary design definitions of prototypes and
test vehicle | Dec' 01 |
| • Parts procurement and fabrication | Apr' 02 |
| • Build test vehicle | Jun' 02 |
| • Cell design and fabrication | Oct' 02 |
| • Build prototype | Mar' 03 |
| • Demonstrate performance and reliability | Jul' 03 |



DEVELOPMENT OF TERRESTRIAL CONCENTRATOR MODULES USING HIGH-EFFICIENCY MULTI-JUNCTION SOLAR CELLS

Funded in Part by NREL/ENTECH Subcontract No. ADJ-1-30620-02

Mark O'Neill

ENTECH, Inc. 1077 Chisolm Trail Keller, TX 76248 USA

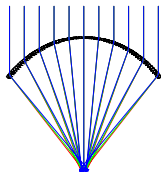
Tel: 817-379-0100 Fax: 817-379-0300

Web: www.entsolar.com E-Mail: mjoneill@entsolar.com



ENTECH, INC.

Slide 1 - High-Performance Kick-Off Meeting at NREL 10/18/01



Background: ENTECH's 4th Generation Module

◆ Fresnel Lens:

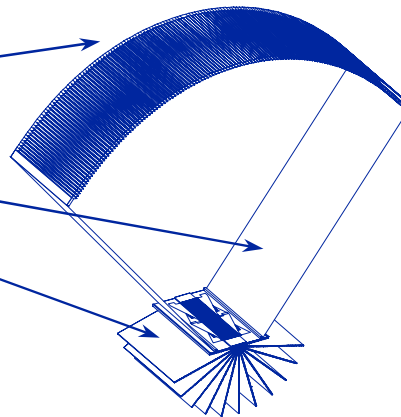
- 3M-Produced Acrylic Lens
- 0.8 m Wide by 3.7 m Long
- Aperture = 3 sq.m. Total

◆ Marine Aluminum Housing

- 2 Sidewalls and 2 Endplates

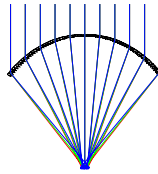
◆ Photovoltaic Receiver:

- 37 Diode-Protected, Prism-Covered Cell Packages
- Full-Receiver Dry Tape & Film Laydown & Encapsulation
- Extruded Aluminum Heat Sink



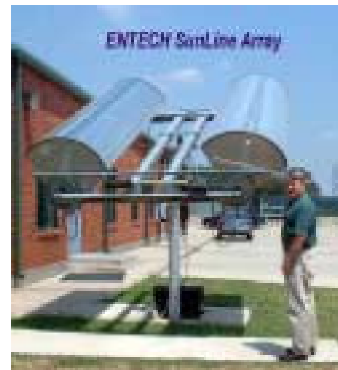
ENTECH, INC.

Slide 2 - High-Performance Kick-Off Meeting at NREL 10/18/01

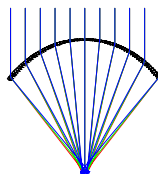


Background: ENTECH's Solar Arrays

Field-Proven 72-Module SolarRow Array for Large Power Plants and 2-Module SunLine Array for Small Applications

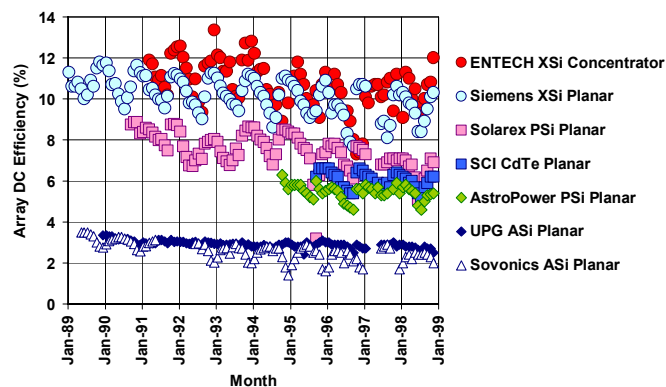


Slide 3 - High-Performance Kick-Off Meeting at NREL 10/18/01

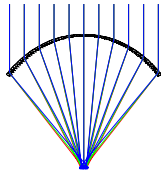


Background: Independent Long-Term Performance Data

PVUSA Array Monthly Efficiency Data at Davis, CA



Slide 4 - High-Performance Kick-Off Meeting at NREL 10/18/01



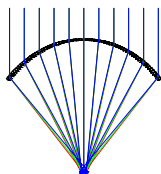
Background: ENTECH Space Concentrators



- ◆ ENTECH Has Been Developing Fresnel Lens Concentrators for Space Power for NASA and DOD Since 1986
- ◆ Most Recently, the Deep Space 1 SCARLET Array Powered Both the Spacecraft and the Ion Engine for the Last 3 Years, Resulting in a Spectacular Comet Rendezvous Sept. 22, 2001
- ◆ Fresnel Lenses Over Triple-Junction Cells Have Performed Flawlessly

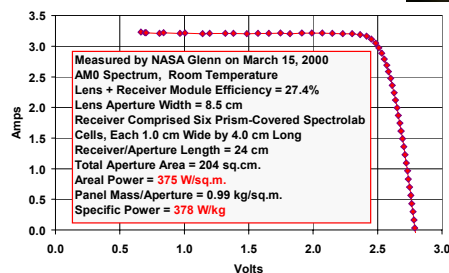


Slide 5 - High-Performance Kick-Off Meeting at NREL 10/18/01



Background: Space Stretched Lens Array (SLA)

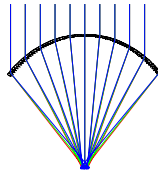
NASA-Measured Performance Is Unprecedented



Ultra-Light Concentrator Array Weighs Much Less per Unit Area than a One-Sun Cell Alone



Slide 6 - High-Performance Kick-Off Meeting at NREL 10/18/01



Synergy Between Space and Ground Concentrators

- ◆ Space Stretched Lens Modules with Space Multi-Junction Cells Perform Spectacularly Outdoors Under Terrestrial Sunlight
- ◆ All Three Modules Below Exceed 27% Operational Efficiency
- ◆ NREL Confirmed Left Module as World-Record Holder

Spectrolab Cell



EMCORE Cell

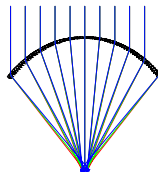


TECSTAR/JX Crystals Cells



ENTECH, INC.

Slide 7 - High-Performance Kick-Off Meeting at NREL 10/18/01



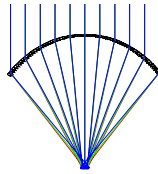
ENTECH's High-Performance Module Development

- ◆ Since 1999, We Have Been Working on the Development of Terrestrial Concentrator Modules Using Multi-Junction Cells
- ◆ Our Proprietary New 440X MJ Cell-Based Module Will Be a Plug-and-Play Replacement for Our Silicon-Based 21X Module, Enabling Use of Proven SunLine and SolarRow Arrays
- ◆ Secondary Optics Will Focus Sunlight in the Longitudinal Direction by Another 21X, Resulting in a 440X Overall Concentration
- ◆ Prototype Testing Indicates that Optical and Thermal Issues Are Technically and Economically Solvable for the New Module
- ◆ However, Cell Issues Remain, But Non-Disclosure Agreements with Cell Suppliers Prevent Open Discussion of Details on These Issues



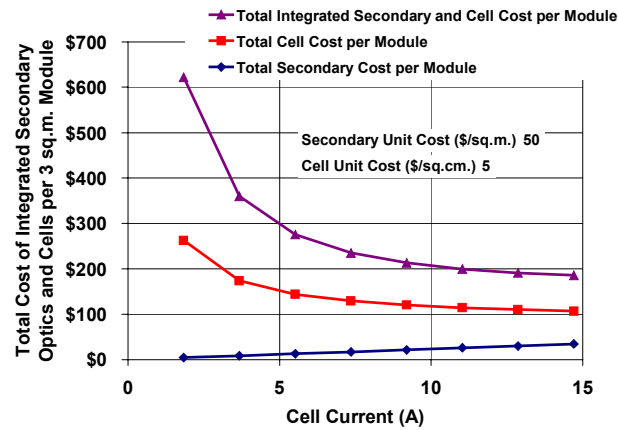
ENTECH, INC.

Slide 8 - High-Performance Kick-Off Meeting at NREL 10/18/01

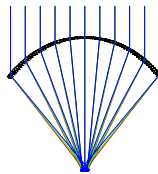


We Need High-Current Cells for Good Economics

**Space Cells
All Operate
Well Under
1 Amp - We
Need Cells
to Operate
Above 10
Amps**

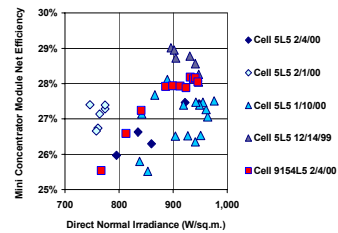


Slide 9 - High-Performance Kick-Off Meeting at NREL 10/18/01



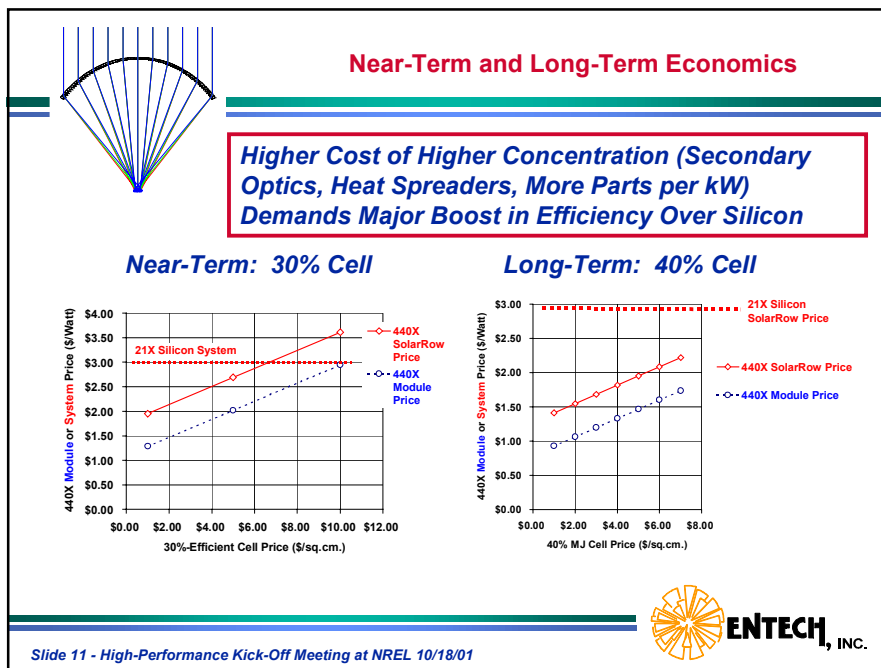
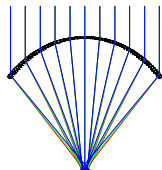
Spectral Issues

- ◆ Cells Designed for AM0 Perform Exceptionally Well in Outdoor Tests Under All Sorts of Conditions
- ◆ Using Quantum Efficiency Curves for AM0-Designed Cells and the AM1.5D Spectrum, the Middle Junction (GaAs) Should Have a 38% Higher Current than the Top Junction (GaInP), Making Our Excellent Outdoor Test Results Impossible
- ◆ This Spectral Problem Is Causing Confusion Among All the Cell Suppliers and Needs to Be Quickly Resolved by Abandoning AM1.5D




Slide 10 - High-Performance Kick-Off Meeting at NREL 10/18/01



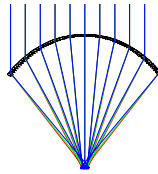



On-Going Work and Key Issues

- ◆ Optimize Operational Current Level for Multi-Junction Cells
 - Is There an Upper Current Limit for Durability?
- ◆ Optimize 440X Module Optics, Including Color-Mixing Lenses
 - Sizing Depends on Current Level Mentioned Above
- ◆ Optimize 440X Module Cells
 - What Spectrum Is Appropriate for Current-Matching the Junctions?
- ◆ Optimize a Robust 440X Photovoltaic Receiver
 - What Caused Recent Boeing and Loral Array Problems on Orbit?
- ◆ Build and Test Full-Size 440X Concentrator Module & SunLine
 - Schedule Depends on Previous Items
- ◆ Commercialize 440X SolarRow and SunLine Systems
 - What Price/sq.cm. Can Multi-Junction Cells Reach?



Slide 12 - High-Performance Kick-Off Meeting at NREL 10/18/01



Recommendations Relating to Multi-Junction Cells

As Concentrator Module Developers and Manufacturers, We Need Help from Cell Suppliers and NREL in Two Key Cell Areas

◆ Solar Spectrum

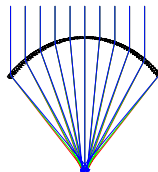
- Near-Term: Cell Suppliers Should Optimize Cells for AM1.5G or AM0, Not AM1.5D, Which Is Not Realistic
- Mid-Term: NREL Should Consider Deploying Triple-Head Pyrheliometers (GaInP, GaInP-Filtered GaAs, and Black Body Sensors) at Various Sites Around the U.S. and Collect Data

◆ Cell Durability

- Near-Term: NREL Should Consider Coordinating Efforts with Spectrolab, EMCORE, TECSTAR, JX, and Other Cell Suppliers to Address Common Issues (Current Limits, Reverse Bias Effects, Transient Effects, Long-Term [30 Years] Durability in Terrestrial Environment, etc.)



Slide 13 - High-Performance Kick-Off Meeting at NREL 10/18/01



Conclusions

- ◆ We Have Already Seen Cell Efficiencies Above 30% and Operational Mini-Module Efficiencies Above 27% in Outdoor Tests of Low-Concentration Space Lenses Over Space Cells
- ◆ These High Efficiencies Could Lead to Outstanding System-Level Economics, If Remaining Issues Are Resolved
- ◆ Significant Cell Issues Remain to Be Addressed:
 - Performance Under High, Non-Uniform Concentration
 - Absolute Current Limits Consistent with 30-Year Life
 - Durability Under Continuously Transient Conditions for 30-Year Life
 - Cell Package Cost, Including Interconnects, Heat Spreader, Bypass Diode, Encapsulation, etc., Must Be \ll \$10/sq.cm. (Current Space)
 - Potential Customers Always Ask for Long-Term Field Data



Slide 14 - High-Performance Kick-Off Meeting at NREL 10/18/01

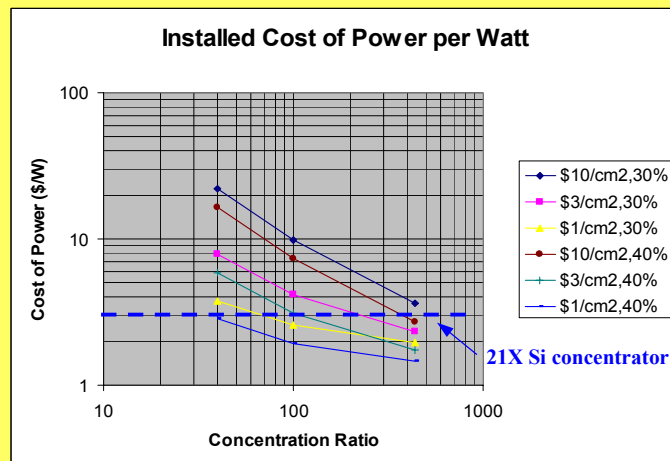
A 3J Solar Cell for High Concentration Terrestrial Applications

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Albuquerque, NM 87123

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Concentrator System Cost Dependencies

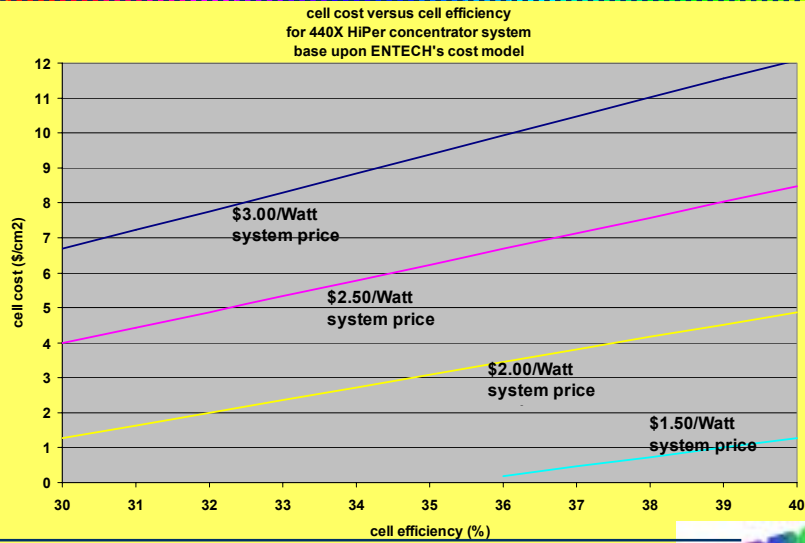


* System model courtesy of M. O'Neill (ENTECH)

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System Cost Dependence of cell cost and efficiency



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Present State-of-the-Art two-terminal MJ concentrators

Embodiment of Concentrator	Supplier	Spectrum	Concentration Ratio	Test Method	Efficiency
L-M InGaP/GaAs/p-Ge (3J)	Spectrolab	AM1.5D	372	Pulsed	32.40%
L-M InGaP/GaAs/n-Ge (2J)	Spectrolab	AM1.5D	160	Pulsed	29.60%
L-MM InGaP/InGaAs/ GaAs (2J)	Fraunhofer	AM1.5D	300	Pulsed	31.30%

* Efficiency numbers assume spectral accuracy

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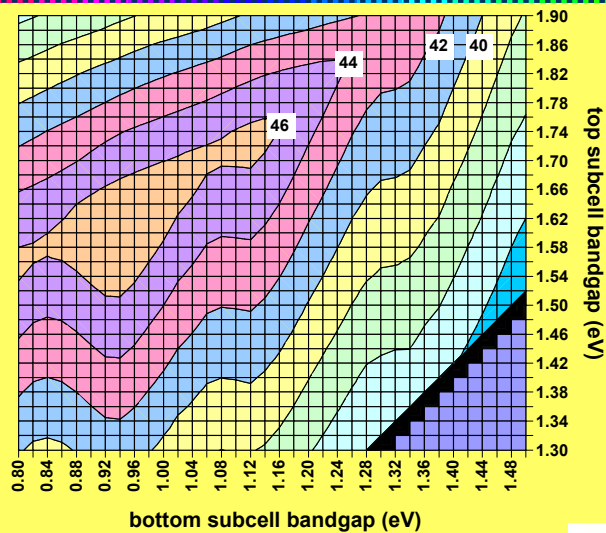
Technical Approach

- Target > 40% cell design for 400X-500X concentration.
- Leverage knowledge of 3J growth on Ge with Lattice-mismatched III-V materials technology presently used by Emcore for Thermophotovoltaic (TPV) devices.
- Device modeling to be performed by Dan Aiken (Emcore) & Prof. Jeffery Gray at Purdue University to determine optimum concentrator cell structure.
- On-sun receiver performance measurements will be made by Amonix.

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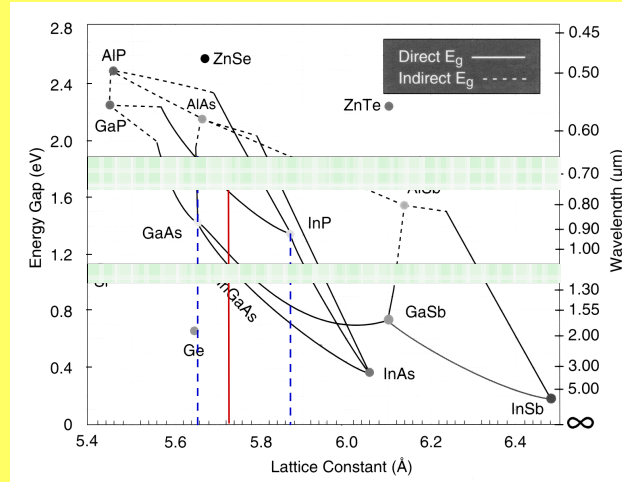
Isoefficiency Plot for 400X Concentration Ratio



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Crystal Growth Chart w/ Optimal Bandgap combinations



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Anticipated Issues for Success of Lattice-Mismatched Cells

- Development of lattice constant grading buffer structure from Ge to $\text{In}_x\text{Ga}_{(1-x)}\text{As}$ ($0 < x < 0.2$)
 - must minimize threading dislocation density
 - Benchmarks on GaAs
 - » 21.6% (AM1.5G) n/p $\text{In}_{0.17}\text{Ga}_{0.83}\text{As}$ (Fraunhofer)
 - » $\text{Voc}=786$ mV, $\text{Jsc}=36.4$ mA/cm², FF=79%
- Development of a high peak current density tunnel diode
 - 500X requires $\text{Jp}=8$ A/cm²
 - Fraunhofer AlGaAs/InGaP diode $\text{Jp}=0.53$ A/cm²
 - Spectrolab high E_g diode did not current limit at 500X

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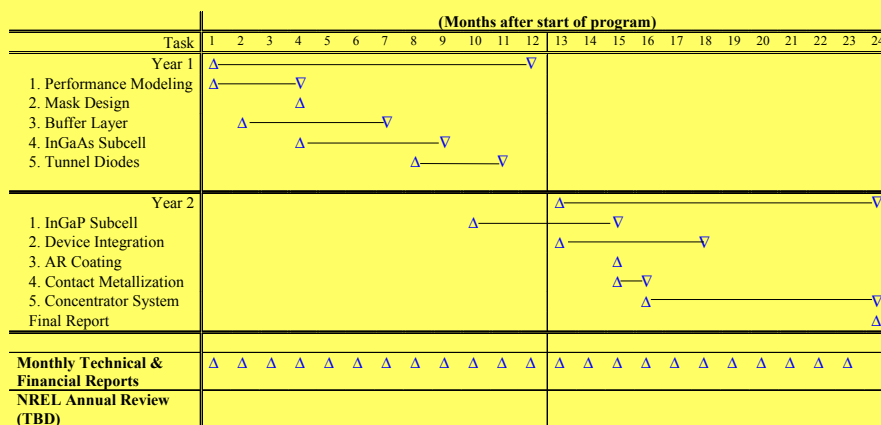
Anticipated Issues for Success of Lattice-Mismatched Cells

- Lattice-matching of top cell to middle cell
 - Tensile strain can cause severe wafer bowing with 140um thick Ge used for space solar applications.
 - High resolution asymmetric bragg x-ray diffraction is necessary for proper matching of solar cell active layers.
- Reproducibility of InGaAs misfit dislocation network
- Impact of misfit/threading dislocations on $\text{In}_x\text{Ga}_{1-x}\text{P}$ bulk properties such as minority carrier lifetime is unknown.

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Project Milestone Chart



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